

# Ab initio Modelling of Thermoelectricity

DFT-based and Semi-Classical Boltzmann Transport Approaches  
with a Focus on the Electronic Properties

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GDR NAME – Ecole MONACOST – 08-13/05/2022

# Content of This Lecture

1. Thermoelectric Effects: Macroscopic View
2. Thermoelectric Coefficients: Microscopic View
3. Transport Equation To Calculate Thermoelectric Coefficients
4. Semi-Classical Approaches To Calculate The TE Coefficients
5. Examples

# References

Some of the slides presented in this lecture are inspired from

Mark LUNDSTROM and Supriyo DATTA

lectures:

**Near-Equilibrium Transport: Fundamentals and Applications**

@ <https://www.youtube.com/watch?v=SddB0DWrc7g>

**Introduction à la thermoélectricité**

@ <https://www.youtube.com/watch?v=2w7NBuu5w9c>

**Fundamental of Nanoelectronics A&B**

@

<https://www.youtube.com/watch?v=p5nsWUKiG9k&list=PLtkeUZItwHK6lvGu8kFKBdhz3XaIZQDFj>

# Content of This Lecture

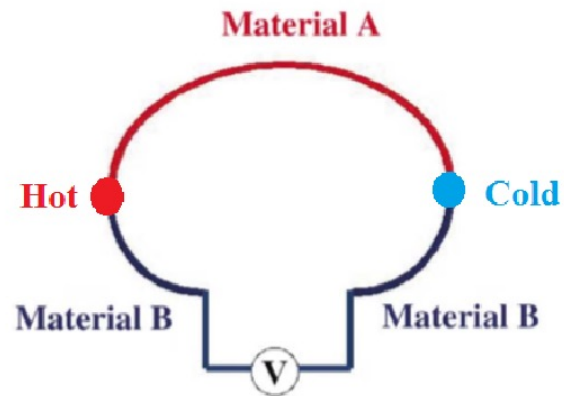
1. Thermoelectric Effects: Macroscopic View
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# Macroscopic View on Thermoelectric Effects

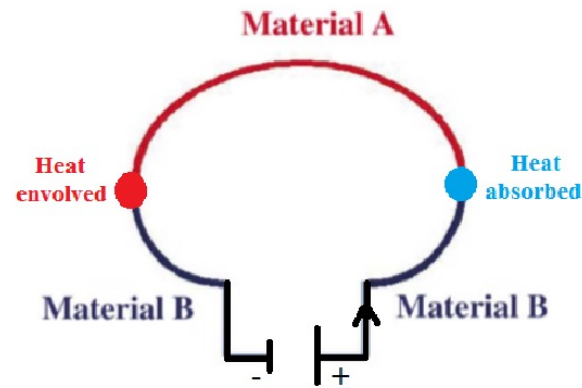
- 1794: Thermoelectric effect first evidenced by Volta
- 1822– 1851: Three effects evidenced

1822: Seebeck effect



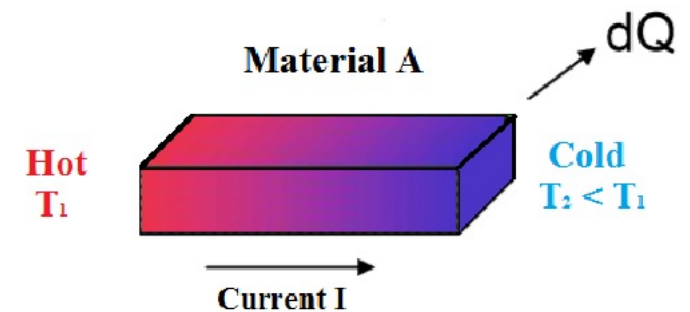
$$S = \frac{\Delta V}{\Delta T}$$

1834: Peltier effect



$$\Pi = \frac{Q}{I}$$

1851: Thomson effect



$$\tau = \frac{dQ}{I\Delta T}$$

# Figure of Merit in Thermoelectricity

- Materials Properties to take into account
  - Electronic properties:
    - Seebeck coefficient:  $S$  ( $\mu\text{V K}^{-1}$ )
    - Electrical conductivity  $\sigma$  ( $\Omega^{-1} \text{m}^{-1}$ )
    - Electrons thermal conductivity  $\kappa_e$  ( $\text{W m}^{-1} \text{K}^{-1}$ )
  - Lattice property:
    - Lattice thermal conductivity (phonons):  $\kappa_L$  ( $\text{W m}^{-1} \text{K}^{-1}$ )
- Properties gathered in one expression: the figure of merit

$$ZT = \frac{S^2 \sigma}{\kappa_e + \kappa_L} \times T$$

( $Z$  in  $\text{K}^{-1}$ ,  $ZT$  adimensional)

# Figure of Merit in Thermoelectricity

$$ZT = \frac{S^2 \sigma}{\kappa_e + \kappa_L} \times T$$

Ideally: ZT should be as large as possible, hence

- Large Seebeck coefficient and electrical conductivity
- Low thermal conductivities

BUT: things are not that easy and TE properties are interrelated:

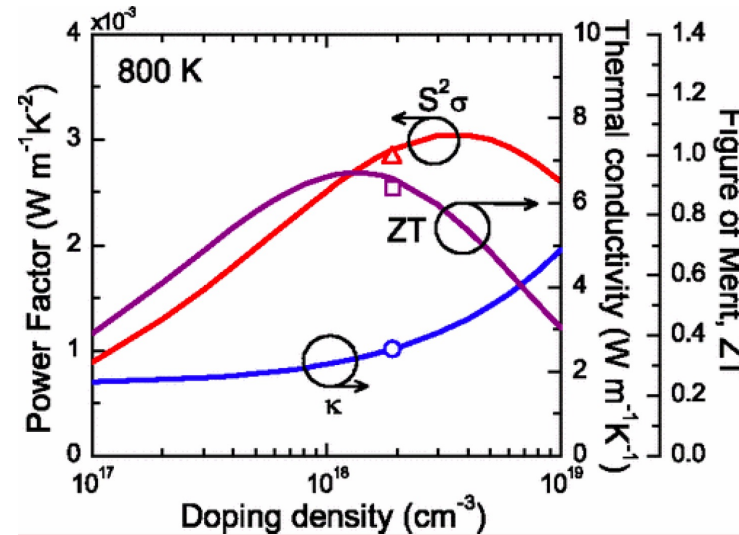
$$S \propto 1/\sigma$$
$$\sigma \propto \kappa_e$$

# Electronic Properties: $S$ , $\sigma$ and $\kappa_e$

Very dependent on:

- The carriers concentration

Pisarenko-type of plot:



Best power factor  $S^2\sigma$  and ZT for moderately doped SC

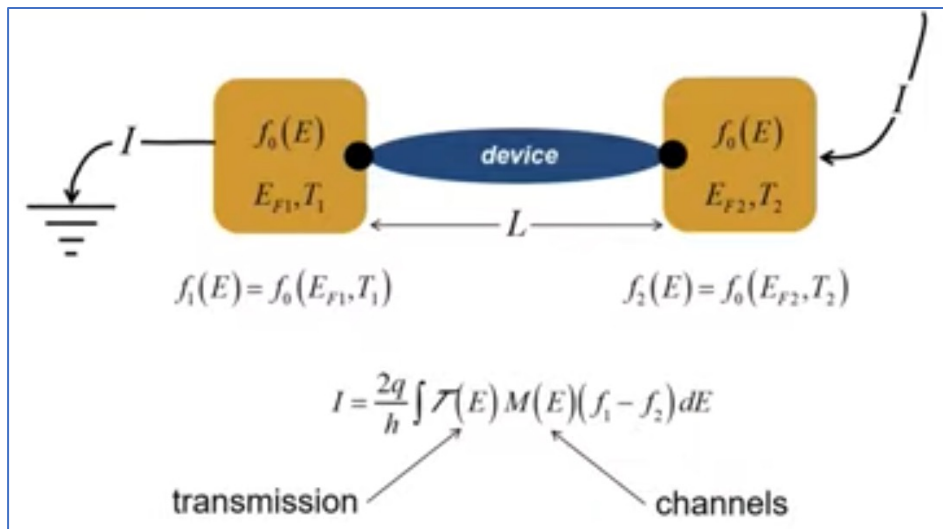
- The band structure  $\Rightarrow$  influence of strains, resonant levels
- Carriers scatterers: e.g., impurities, defects

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# Microscopic View of The Electrical Conductivity

- We take the view of the *Landauer approach* ( $\neq$  the Boltzmann one)
- Start with the electrical conductivity



- A device of length  $L$  is connected to two contacts
- The one on the left is grounded
- The current flow from right to left

Difference  $f_1 - f_2$  due to difference in chemical potentials  
Current is proportional to:

- So:  $f_1 - f_2 \approx -\left(\frac{\partial f_0}{\partial E}\right)(q\Delta V)$
- Transmission coefficient ( $0 \leq T \leq 1$  semi-classical approach)
- Number of channels (prop. to carriers velocity  $\times$  DOS)
- Difference in Fermi functions caused by a difference in the Fermi levels (electrochemical potential)

# Microscopic View of The Electrical Conductivity

- Expressions for  $J$  and  $\sigma$ :

$$J_x = \sigma_n \frac{d(F_n/q)}{dx} \rightarrow$$

$$\sigma_n = \frac{2q^2}{h} \int \lambda(E) \frac{M(E)}{A} \left( -\frac{\partial f_0}{\partial E} \right) dE$$

- $F_n$  is the *quasi-Fermi level* that varies across the device (or material) due to the applied voltage ( $F_n/q$  = electrochemical potential)
- $\lambda(E)$  is the *mean-free path* of electrons with energy  $E$ ,  $A$  is the cross section
- $f_0$  is the *Fermi-Dirac distribution* at equilibrium: approximate  $f_1 - f_2$

# Microscopic View of The Electrical Conductivity

Recast the electrical conductivity expression as

$$\sigma = \int \sigma'(E) dE$$

with:

$$\sigma' = \frac{2q^2}{h} \lambda(E) \frac{M(E)}{A} \left( -\frac{\partial f_0}{\partial E} \right) = q^2 \Xi(E) \left( -\frac{\partial f_0}{\partial E} \right) \text{ the } \textit{differential electrical conductivity}, \text{ and}$$

$$\Xi(E) = \frac{2}{h} \frac{M(E)}{A} \lambda(E) \text{ called the } \textit{transport function}$$

The transport function is hence prop. to the mean-free path of the electrons and to the number of conduction channels

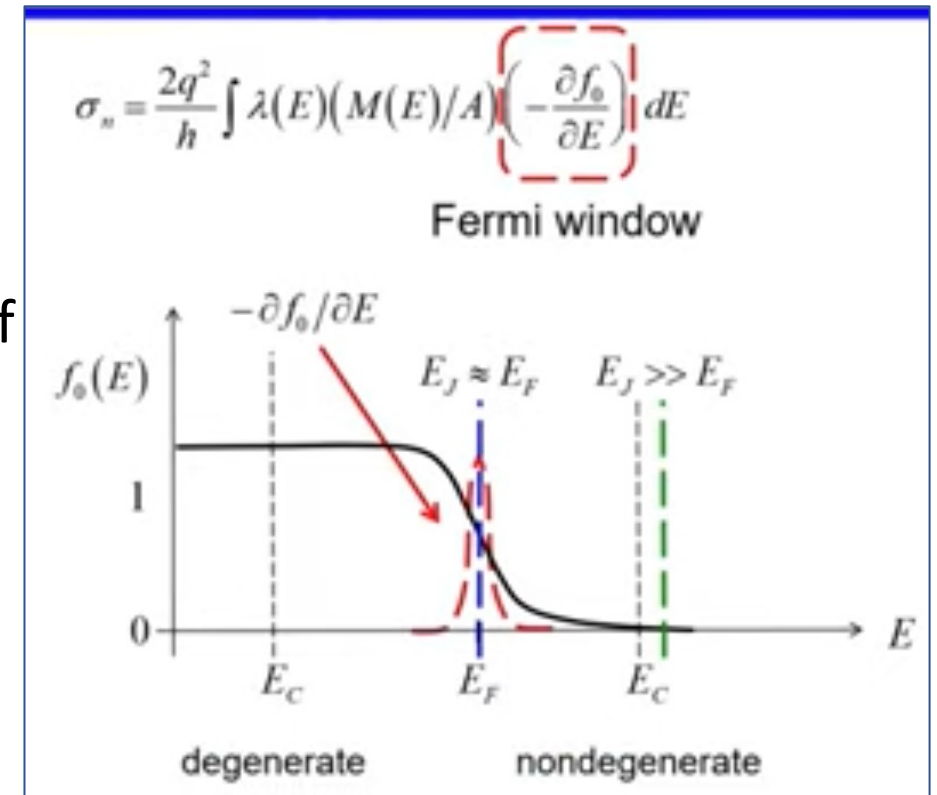


# Microscopic View of The Electrical Conductivity

At what energy does the current flow? In the conduction band, but where?

Dictated by the *Fermi Window*

- For a non-degenerate SC:  $E_F \ll E_J$   
Only the tail of the FW is responsible for the conduction that occurs close to the bottom of the CB
- For a degenerate SC:  $E_F \approx E_J$   
The current is flowing near the Fermi energy where the FW is. This occurs for metal of heavily doped SC



# Microscopic View of The Seebeck Coefficient

- Recall: the Seebeck coefficient is *responsible for the current to flow when a temperature difference is applied*
- Seebeck coefficient also called the *thermopower*
- Temperature difference causes electrons to flow because of a *difference between the Fermi functions of the two contacts*

$$I = \frac{2q}{h} \int T(E)M(E)[f_1(E) - f_2(E)] dE$$

But now the difference  $f_1 - f_2$  is caused by a *difference in temperatures*

So, now:  $f_1 - f_2 \approx -\frac{\partial f_0}{\partial T} \Delta T \propto -\frac{\partial f_0}{\partial E} \Delta T$  (See: <https://www.youtube.com/watch?v=5iTehoVGYeE>)

# Microscopic View of The Seebeck Coefficient

The electrical current becomes

$$I = - \left\{ \int \left( \frac{E - E_F}{qT} \right) \sigma'(E) dE \right\} \Delta T = (SG) \Delta T$$

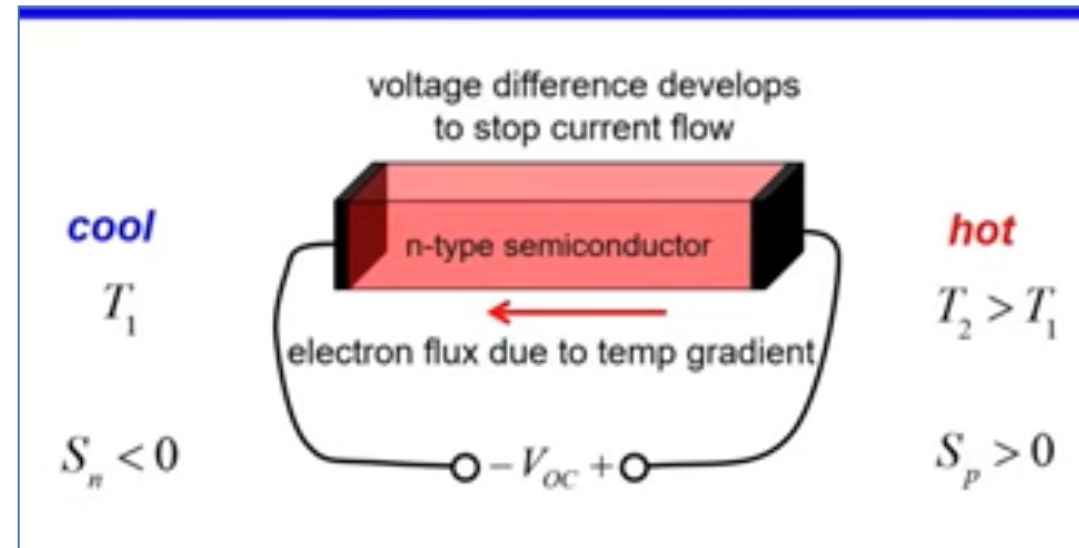
$$J_x = -S\sigma \frac{dT}{dx}$$

$$S\sigma = - \int \left( \frac{E - E_F}{qT} \right) \sigma'(E) dE \rightarrow S = - \frac{\int \left( \frac{E - E_F}{qT} \right) \sigma'(E) dE}{\int \sigma'(E) dE}$$

# Microscopic View of The Seebeck Coefficient

- What is the sign of the Seebeck coefficient?

$J = \sigma \mathcal{E}$ (voltage gradient only)	$J_s = -S\sigma \frac{dT}{dx}$ (temperature gradient only)
$J_s = -\sigma \frac{dV}{dx} - S\sigma \frac{dT}{dx}$	<div style="border: 1px solid red; padding: 5px; display: inline-block;"><math>V_{oc} = -S\Delta T</math> <math>V_{oc} = V_2 - V_1</math> <math>\Delta T = T_2 - T_1</math></div>
$J_s = 0$	
measure open-circuit voltage	

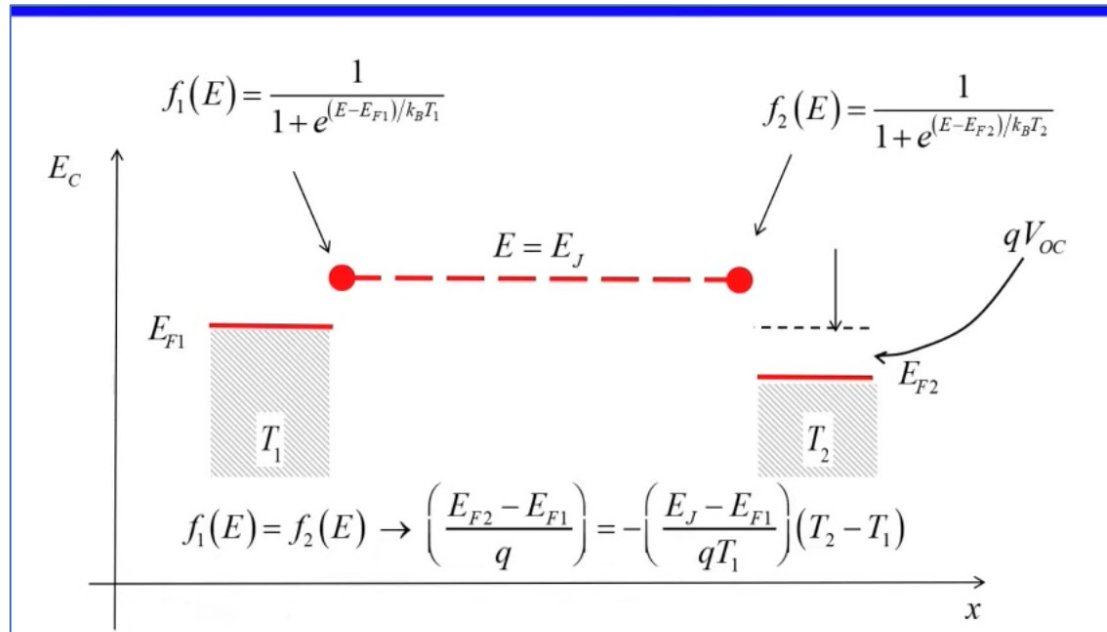


- We are dealing with a n-type SC
- Hot side on the right / cool side on the left
- Electrons diffuse from hot side to cool side

- A positive voltage appears to drag the electrons back to counteract the diffusion and give a zero current under open circuit voltage ( $V_{oc}$ )
- Because  $V_{oc}$  is  $-S\Delta T$  (Seebeck coefficient) we conclude That  $S$  is negative for n-type SC

# Microscopic View of The Seebeck Coefficient

## Simple meaning and expression of the Seebeck coefficient



- $E_J$ : Average energy at which current flows near the bottom of the conduction band in a non-degenerate n-type SC
- In OC conditions:  $f_1 = f_2$
- $V_{OC} = -S\Delta T$

$$V_{OC} = \left( \frac{E_J - E_F}{-qT_1} \right) \Delta T$$

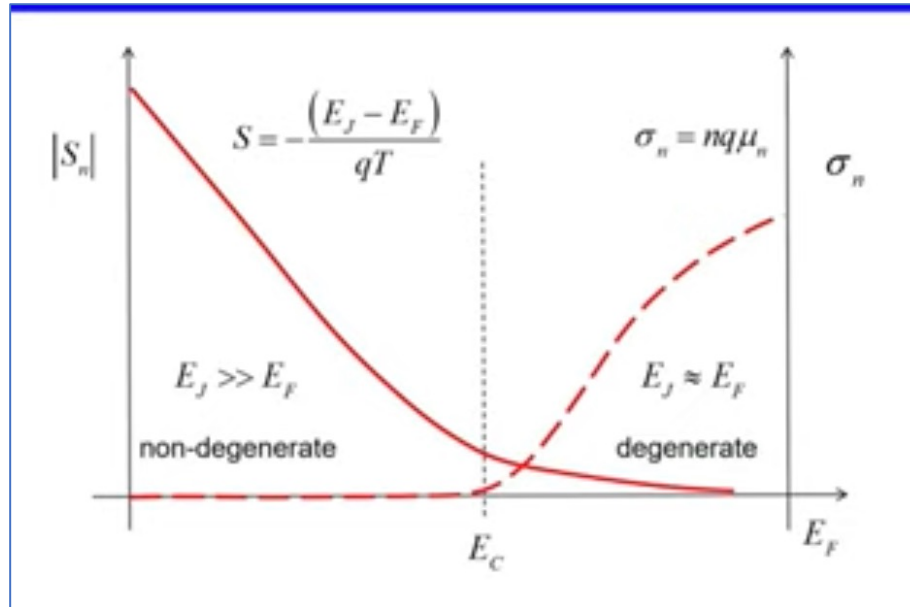
$$\text{and } S = - \frac{\int \left( \frac{E - E_F}{qT} \right) \sigma'(E) dE}{\int \sigma'(E) dE}$$

Average expression:  $\left\langle \frac{E - E_F}{qT} \right\rangle$

Hence:

The Seebeck coefficient is *related to the average energy at which the current flows with respect to the Fermi energy*

# Seebeck coefficient Versus Electrical Conductivity



- If the Fermi level approaches the level at which conduction occurs,  $S$  decreases
- *If the "gap" is large,  $S$  is large too*
- When the Fermi level gets close to the level at which conduction occurs,  $\sigma$  increases
- *If the "gap" is small,  $\sigma$  is large*

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# How to Calculate These Coefficients?

The TE coefficients = transport coefficients hence  
Need for a transport equation



Boltzmann transport equation

If no need to account for strong e-e quantum correlation?



Mean-field approach (e.g. DFT) +  
Semi-classical considerations +  
Relaxation time approximation



# The Boltzmann Transport Equation

$$\frac{df}{dt} = 0$$

$f$ : the distribution of electrons (Fermi-Dirac)  $\rightarrow$  electronic props.  
or phonons (Bose-Einstein)  $\rightarrow$  thermal-related props.

$f$  is 6-dimensional function that evolves with time:  $f(r, p, t)$

$$\frac{\partial f}{\partial t} + \vec{v} \cdot \nabla_r f + \vec{F}_e \cdot \nabla_p f = 0$$

(no collision processes accounted for here)

# The Boltzmann Transport Equation

If one accounts for collisions:

$$\frac{\partial f}{\partial t} + \vec{v} \cdot \nabla_r f + \vec{F}_e \cdot \nabla_p f = \left. \frac{df}{dt} \right|_{coll} = \hat{C}f = G_{in} - G_{out}$$
$$\hat{C}f = -\frac{f(p) - f_0(p)}{\tau_m} = -\frac{\delta f}{\tau_m}$$

with  $\delta f$  a small variation of the distribution function with respect to equilibrium ( $f_0(p)$ ) and  $\tau_m$  the scattering time of the carriers.

*⇒ Relaxation Time Approximation.*

- 1) RTA not valid in general! Justified when isotropic and/or elastic scatt.
- 2) Widely used in practice, though.

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# Tools to Calculate Thermoelectric Properties

- Electronic structures  $\Rightarrow$  quantum ab initio approaches
- Electronic TE properties: Seebeck coefficient, electrical and electronic thermal conductivities and
- Lattice TE property: thermal conductivity

$\Rightarrow$  at least apply semi-classical approaches to the *BTE*

# Electronic Structure Calculation

- Any code can do the job: VASP, Quantum Espresso, Wien2k, SIESTA, CP2K,...

```
SYSTEM = PbSnTeSe
PREC = Normal
ISYM = 0
ISTART = 0; ICHARG = 2
ENCUT = 300
ISMEAR = 0; SIGMA = 0.02
INIMIX = 1; IMIX = 1; AMIX = 0.2; BMIX = 0.001
GGA = PZ
LREAL = Auto
##### SCF SETTING #####
IBRION = -1; ISIF = 2
NSW = 0
EDIFF = 1e-7
ALGO = Damped; IALGO = 53; TIME = 0.4
#####
NCORE = 8; KPAR = 8
```

```
&CONTROL
  title          = 'Pb2Bi2Te5',
  calculation     = 'scf',
  nstep          = 50,
  etot_conv_thr  = 1d-7,
  restart_mode   = 'from_scratch',
  verbosity      = 'high',
  prefix         = '225.opt',
  pseudo_dir     = '/ccc/work/cont003/gen6881/gen6881/packages/qe/PSP'
  wf_collect     = .true.,
/
&SYSTEM
 ibrav          = 0,
  celldm(1)     = 1.000000000000d0,
  nat           = 9,
  ntyp          = 3,
  ecutwfc       = 50.d0,
  ecutrho       = 500.d0,
  occupations   = 'fixed',
  nbnd          = 100,
  noncolin      = .true.,
  lspinorb      = .true.,
/
&ELECTRONS
  conv_thr      = 1d-10,
  mixing_beta   = 0.2d0,
  startingpot   = 'atomic',
  startingwfc   = 'atomic+random',
/
ATOMIC_SPECIES
Pb 107. Pb.rel-pbe-dn-rrkjus_psl.1.0.0.UPF
Bi  209. Bi.rel-pbe-dn-rrkjus_psl.1.0.0.UPF
Te 128. Te.rel-pbe-dn-rrkjus_psl.1.0.0.UPF

K_POINTS automatic
4 4 2 1 1 1

CELL_PARAMETERSalat
8.510135436547301d0 -1.215303952543845d-13 1.797830763980544d-20
-4.255067718345414d0 7.369993477081541d0 -1.218922858649270d-20
7.308332310332742d-20 -2.398019053344136d-20 35.41739576994752d0
```

```
PbSnTeSe
P 64 1 P1
RELA
24.831001 24.831001 24.831001 90.000000 90.000000 90.000000
ATOM -1: X=0.98683803 Y=0.02333037 Z=0.75873896
MULT= 1 ISPLIT= 8
Se NPT= 781 R0=0.00005000 RMT= 2.5000 Z: 34.00000
1.0000000 0.0000000 0.0000000
0.0000000 1.0000000 0.0000000
0.0000000 0.0000000 1.0000000
ATOM -2: X=0.48483973 Y=0.00176046 Z=0.76309299
MULT= 1 ISPLIT= 8
Se NPT= 781 R0=0.00005000 RMT= 2.5000 Z: 34.00000
1.0000000 0.0000000 0.0000000
0.0000000 1.0000000 0.0000000
0.0000000 0.0000000 1.0000000
ATOM -3: X=0.99362382 Y=0.52076864 Z=0.76440142
MULT= 1 ISPLIT= 8
Se NPT= 781 R0=0.00005000 RMT= 2.5000 Z: 34.00000
1.0000000 0.0000000 0.0000000
0.0000000 1.0000000 0.0000000
0.0000000 0.0000000 1.0000000
ATOM -4: X=0.48399564 Y=0.50343080 Z=0.75407842
MULT= 1 ISPLIT= 8
Te NPT= 781 R0=0.00001000 RMT= 2.5000 Z: 52.00000
1.0000000 0.0000000 0.0000000
0.0000000 1.0000000 0.0000000
0.0000000 0.0000000 1.0000000
ATOM -5: X=0.99095228 Y=0.01007940 Z=0.25648293
MULT= 1 ISPLIT= 8
Te NPT= 781 R0=0.00001000 RMT= 2.5000 Z: 52.00000
1.0000000 0.0000000 0.0000000
0.0000000 1.0000000 0.0000000
0.0000000 0.0000000 1.0000000
ATOM -6: X=0.49533243 Y=0.00570859 Z=0.26231032
MULT= 1 ISPLIT= 8
Te NPT= 781 R0=0.00001000 RMT= 2.5000 Z: 52.00000
1.0000000 0.0000000 0.0000000
0.0000000 1.0000000 0.0000000
0.0000000 0.0000000 1.0000000
ATOM -7: X=0.98727580 Y=0.50993181 Z=0.25765265
MULT= 1 ISPLIT= 8
Te NPT= 781 R0=0.00001000 RMT= 2.5000 Z: 52.00000
1.0000000 0.0000000 0.0000000
```

As far as their output data can be processed by other programs: e.g. BoltzTraP, phonopy, phono3py, etc.

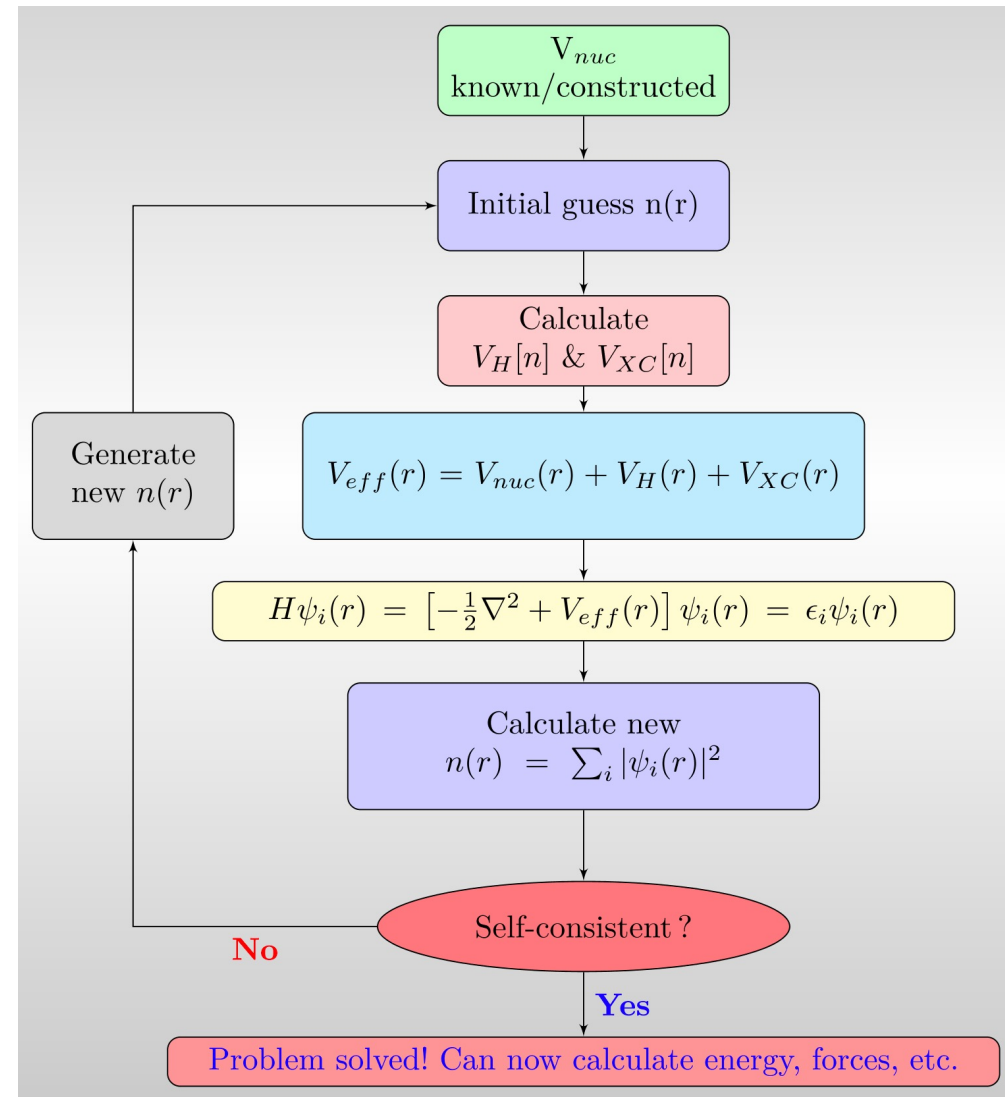
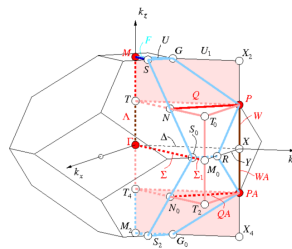
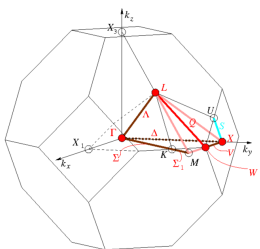
# Electronic Structure Calculation

Reminder on how to solve the Kohn-Sham equations of DFT →

Allows one to obtain:

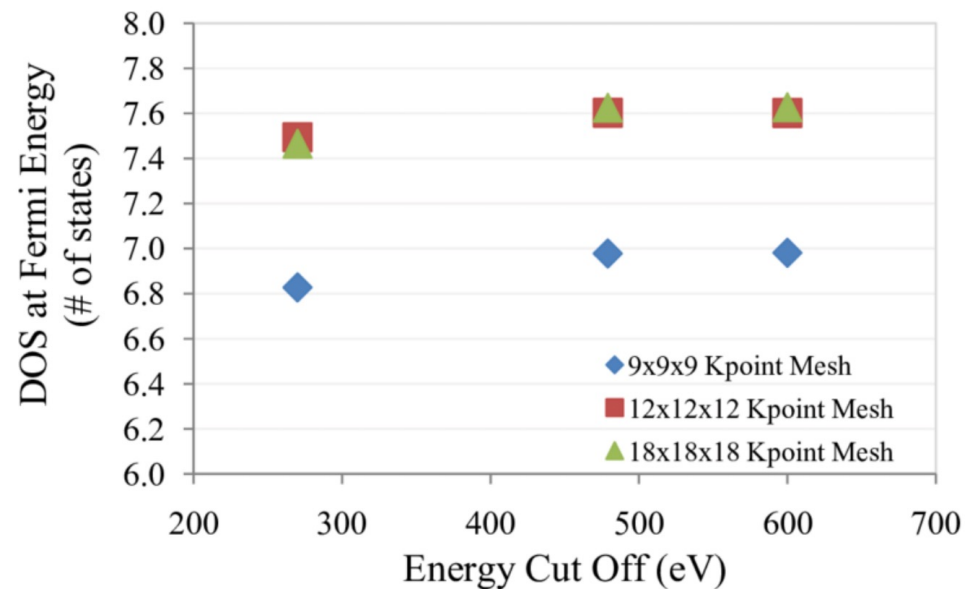
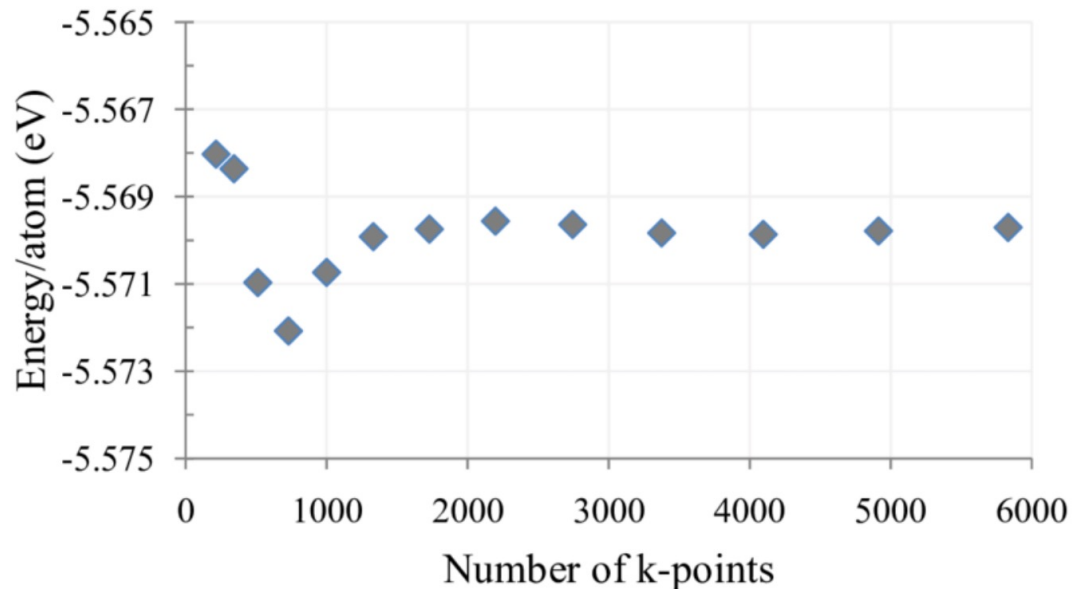
- Electrons energies
- orbitals

for a set of k-points spread over the Brillouin zone



# Convergence With k-Points And Cutoff Energy

- K-points grid: used to sample the Brillouin zone  
*Small lattice* → large reciprocal lattice ⇒ *lot of k-points needed*
- Cutoff energy: defines the plane wave basis set size



# How To Solve The Boltzmann Equation

$$\frac{\partial f}{\partial t} + \vec{v} \cdot \nabla_r f + \vec{F}_e \cdot \nabla_p f = -\frac{\delta f}{\tau_m}$$

Linearize the equation: analytically develop expressions for the transport coefficients expressions (RTA) and calculate them numerically

For electronic coefficients → *BoltzTraP*(2) Code (Boltzmann Transport Program)

(GNU, <https://www.imc.tuwien.ac.at/index.php?id=21094>)



# Theory Behind BoltzTraP

Linearization of the Boltzmann equation

$$\frac{\partial f}{\partial t} + \vec{v} \cdot \nabla_r f + \vec{F}_e \cdot \nabla_p f = -\frac{\delta f}{\tau_m}$$
$$f = f^{(0)}(E) + f^{(1)}(E) = \frac{1}{1 + e^{\frac{E-\mu}{kT}}} + f^{(1)}(E)$$

Recall that  $E = \epsilon(k)$

$$\frac{\partial f^{(1)}}{\partial t} + v \left( \frac{E - \mu}{T} \nabla_r T + \nabla_r \mu - q\mathcal{E} \right) \left( -\frac{\partial f^{(0)}}{\partial \mu} \right) + \frac{q}{\hbar c} (v \times B) \nabla_k f^{(1)} = -\frac{f^{(1)}}{\tau_m}$$

Note:  $\mu$  is the (electro-)chemical potential

$\tau_m$  depends on  $k$

# Theory Behind BoltzTraP

- Assume  $\epsilon_k = \frac{\hbar^2 k^2}{2m^*}$  and  $v_k = \frac{\hbar k}{m^*}$

$$f^{(1)} = \tau(\epsilon) v \left[ (q\mathcal{E} - \nabla\mu) - \frac{\epsilon - \mu}{T} \nabla T \right] \left( -\frac{\partial f^{(0)}}{\partial \epsilon_k} \right)$$

Kinetic coefficient:

$$K_p = \frac{1}{3} \int v^2 (\epsilon - \mu)^p n(\epsilon) \tau(\epsilon) \left( \frac{\partial f^{(0)}}{\partial \epsilon} \right) d\epsilon, \quad p = 0, 1, 2$$

Electrical current:

$$J = K_0 (q\mathcal{E} - \nabla\mu) - K_1 \frac{\nabla T}{T}$$

Electronic thermal current:

$$J_Q = K_1 (q\mathcal{E} - \nabla\mu) - K_2 \frac{\nabla T}{T}$$

# Theory Behind BoltzTraP

- Kinetic coefficient:

$$K_p = \frac{1}{3} \int v^2 (\epsilon - \mu)^p n(\epsilon) \tau(\epsilon) \left( \frac{\partial f^{(0)}}{\partial \epsilon} \right) d\epsilon, \quad p = 0, 1, 2$$

- Electrical conductivity:

$$\sigma = q^2 K_0$$

- Seebeck coefficient:

$$S = \frac{1}{qT} \frac{K_1}{K_0}$$

- Electronic thermal conductivity:

$$\kappa_e = \frac{K_0 K_2 - K_1^2}{K_0 T}$$

Note:  $\sigma$ ,  $S$  and  $\kappa_e$  are tensors, so BoltzTraP outputs tensors

# Practical Equations Calculated in BoltzTraP

- Velocity vector:

$$v_{\alpha}(i, k) = \frac{1}{\hbar} \frac{\partial \epsilon_{ik}}{\partial k_{\alpha}}$$

- Conductivity tensor expressions:

$$\sigma_{\alpha\beta}(i, k) = q^2 \tau_{ik} v_{\alpha}(i, k) v_{\beta}(i, k)$$

$$\sigma_{\alpha\beta}(\epsilon) = \frac{1}{N_k} \sum \sigma_{\alpha\beta}(i, k) \frac{\partial(\epsilon - \epsilon_{ik})}{\partial \epsilon}$$

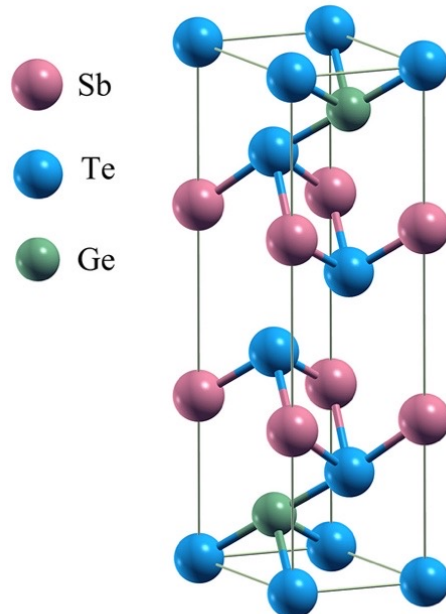
$$\sigma_{\alpha\beta}(T, \mu) = \frac{1}{\Omega} \int \sigma_{\alpha\beta}(\epsilon) \left( -\frac{\partial f(T, \epsilon)}{\partial \epsilon} \right) d\epsilon$$

# A Note On The Convergence With k-Point Grids

How many k-points should I use for a transport calculation?

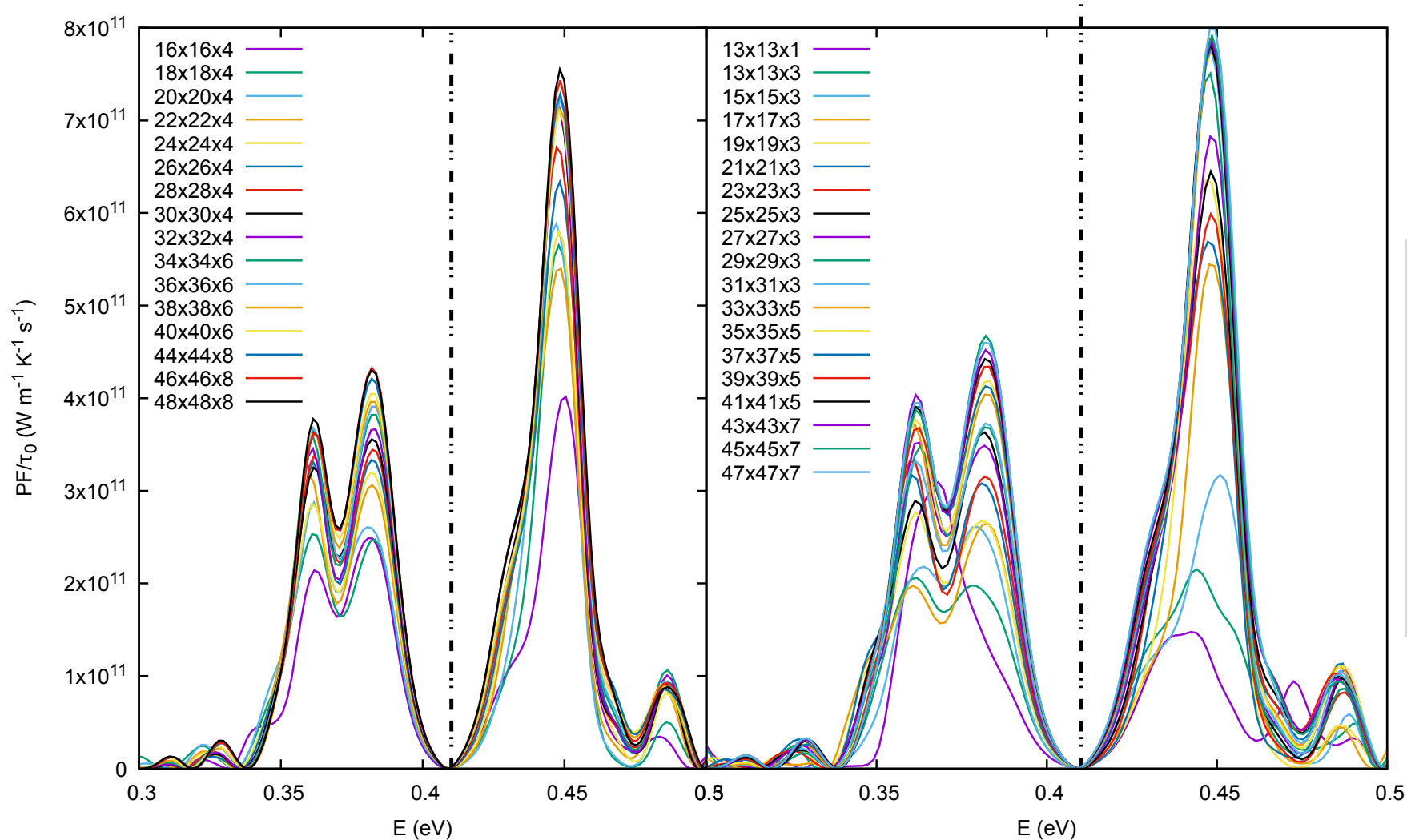
- According to BoltzTraP2 authors: *“a lot”*
- As a rule of thumb: convergence tests should start at  *$16 \times 10^6$  k-points in the IBZ*

Test on  $\text{Pb}_2\text{Sb}_2\text{Te}_5$ :



Hexagonal structure  
 $4.23 \times 4.23 \times 16.92 \text{ \AA}$

# A Note On The Convergence With k-Point Grids



## Comments

- Rather slow convergence with k-point grid
- Same convergence rate with both odd and even grid
- Faster calculations with odd grid than with even one

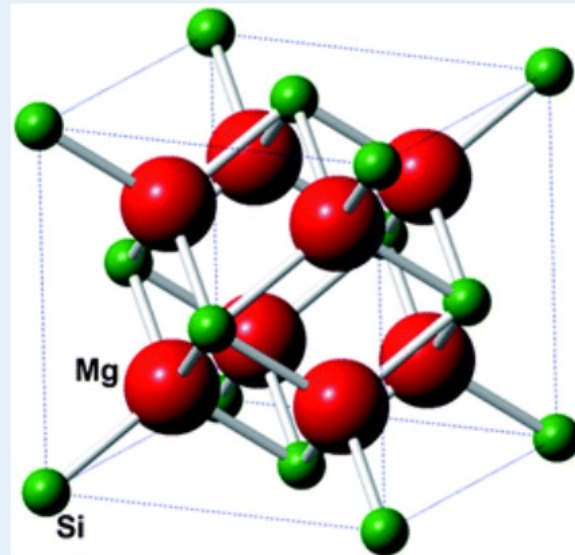
Investigation performed by J. Tian,  
PhD student

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# A Simple Silicide Compound: $\text{Mg}_2\text{Si}$

- Cubic anti-fluorite structure ;  $a=0.635$  nm.
- Si atoms occupy the 4a (0, 0, 0) sites
- Mg atoms occupy the 8c (0.25, 0.25, 0.25) sites.
- The  $Fm\bar{3}m$  space group fixes the fractional coordinates of all atoms.



References:

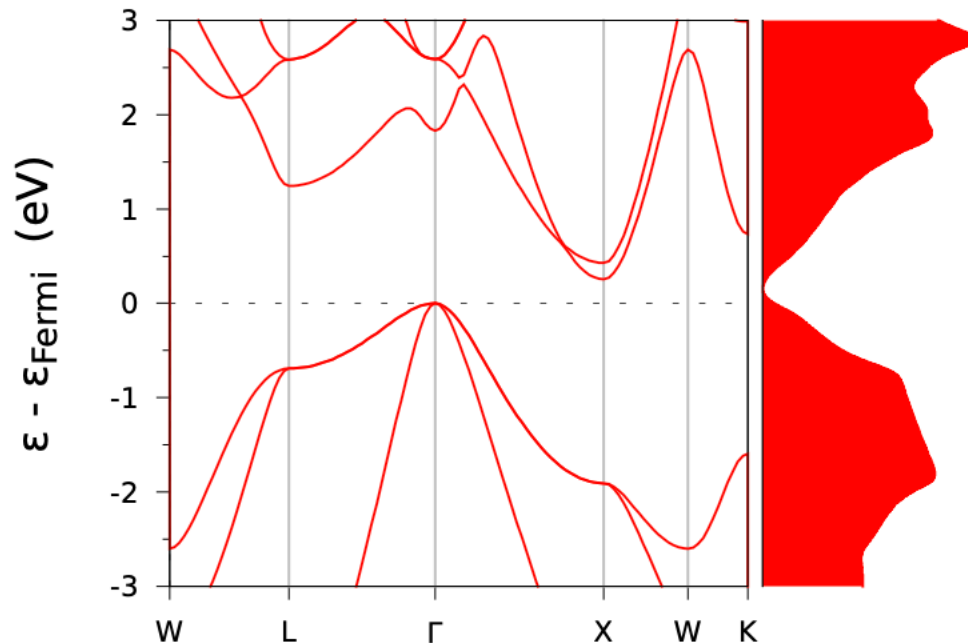
Studies by H. Balout et al.

- J. Mol. Model., 2017 (23) 130
- Eur. Phys. J. B, 2015 (88) 209
- J. Electron. Mater., 2014 (43) 3801
- Intermetallics, 2014 (50) 8
- J. Electron. Mater., 2013 (42) 3458



# Band Structure of Mg<sub>2</sub>Si

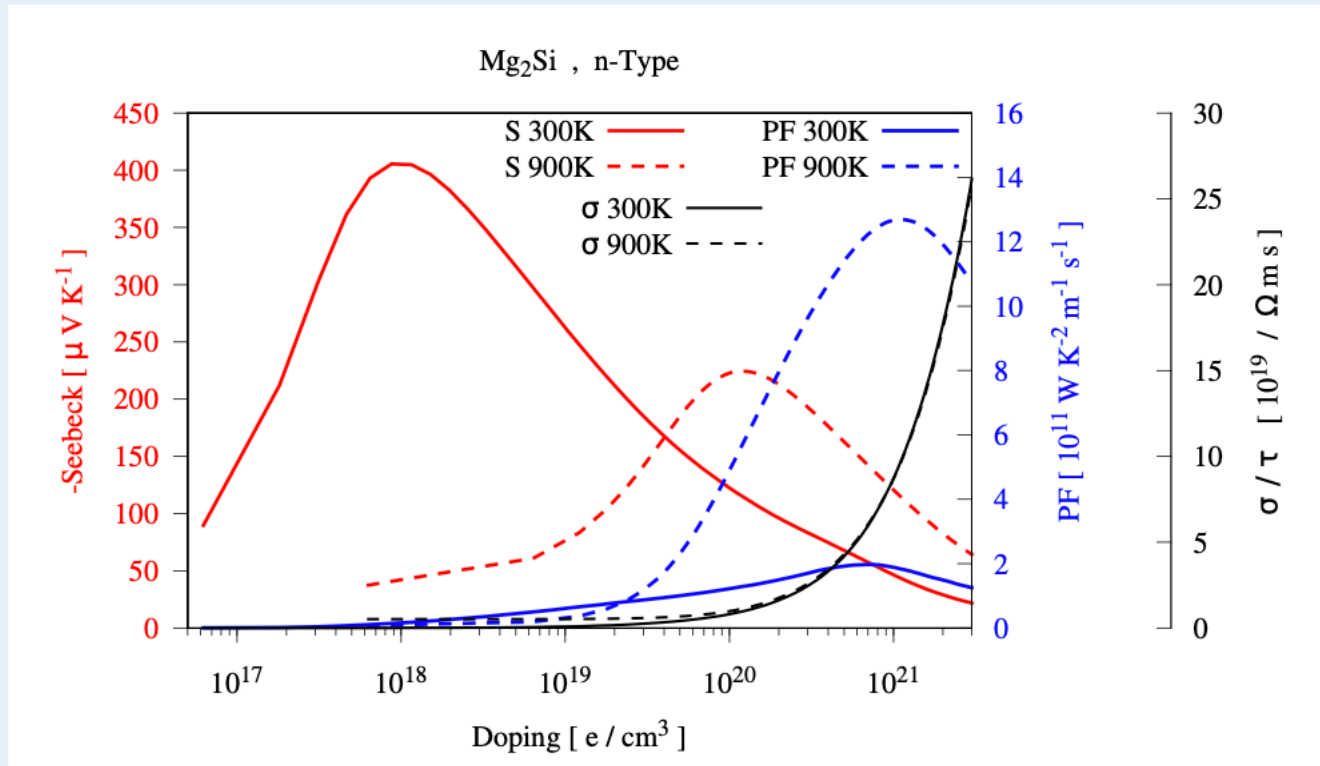
- **Mg<sub>2</sub>Si** Is a **Semiconductor** with Narrow Band Gap.
- Indirect Band Gap  $E_{\Gamma-X} = 256 \text{ meV}$ .
- **Conduction Band** Bears Two Sets of Low-Lying, 3-Fold Degenerated Orbitals **Separated by 173 meV**



- Calculated with the GGA-PBE functional
- Exp.  $\sim 0.7\text{-}0.8 \text{ eV}$

# Transport Coefficients of Bulk n-type $\text{Mg}_2\text{Si}$

- At 300 K and  $\sim 10^{18} \text{ e/cm}^3 \rightarrow S_{\text{max}} = 400 \mu\text{V/K}$
- At 900 K and  $\sim 1.2 \times 10^{20} \text{ e/cm}^3 \rightarrow S_{\text{max}} = 225 \mu\text{V/K}$
- $\sigma/\tau$  Increases dramatically  $\Rightarrow$  Doped Semiconductor.
- $\text{PF}_{\text{max}}$  is Observed at around  $10^{21} \text{ e/cm}^3$ .



## Comments

- $S$  is high for lightly doped  $\text{Mg}_2\text{Si}$
- $\sigma$  is high for heavily doped  $\text{Mg}_2\text{Si}$

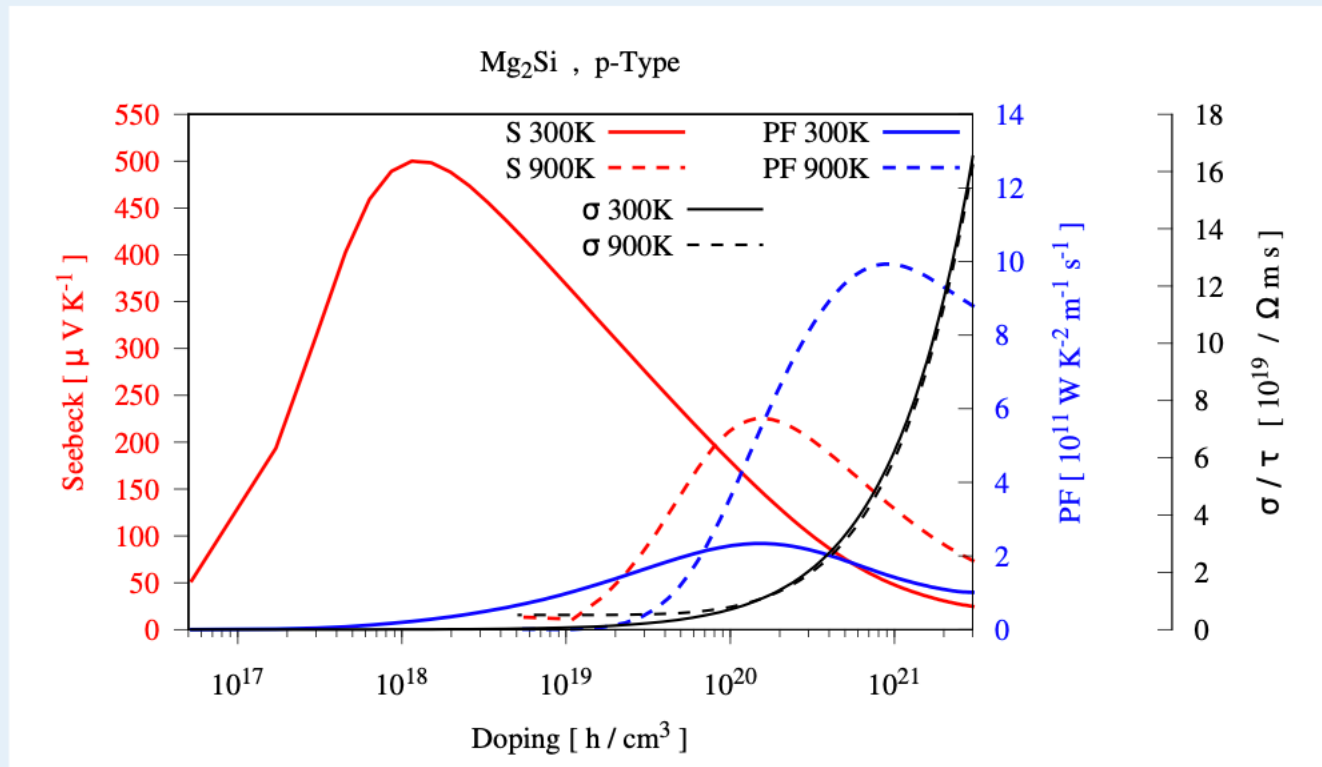
- $S$  and  $\text{PF}$  decrease as  $T$  increases
- $\sigma$  not much sensitive to  $T$

## Note

$\sigma$  and  $\text{PF}$  in units of per second because  $\tau_m$  is unknown

# Transport Coefficients of Bulk p-type Mg<sub>2</sub>Si

- At 300 K and  $\sim 10^{18} \text{ h/cm}^3 \rightarrow S_{\text{max}} = 500 \mu\text{V/K}$
- At 900 K and  $\sim 1.2 \times 10^{20} \text{ h/cm}^3 \rightarrow S_{\text{max}} = 225 \mu\text{V/K}$
- $\sigma/\tau$  Increases dramatically  $\Rightarrow$  Doped Semiconductor.
- $\text{PF}_{\text{max}}$  is Observed in the Range  $10^{20} - 10^{21} \text{ h/cm}^3$ .



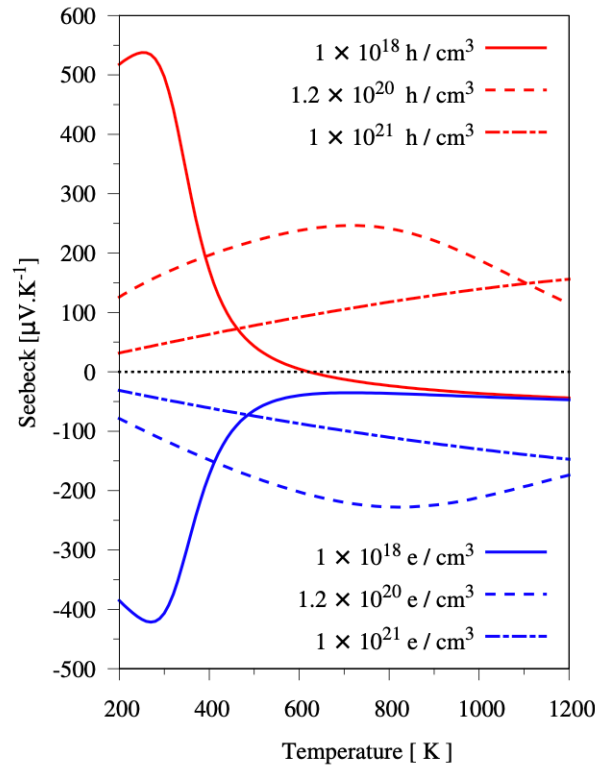
## Comments

Same observations as for n-type doping but

- Seebeck coefficient slightly larger for the same T and doping: 500  $\mu\text{V/K}$  at 300 K and  $10^{18} \text{ h/cm}^3$  instead of 400  $\mu\text{V/K}$
- $\sigma$  slightly smaller for the same T and doping

# Transport Coefficients w.r.t. Temperature

- $S_{max}$   $\begin{cases} \text{n-type, } S = -450 \mu\text{V/K} \\ \text{p-type, } S = 540 \mu\text{V/K} \end{cases}$
- $T \nearrow \Rightarrow S \searrow$  and  $\rightarrow -50 \mu\text{V/K}$  (bipolar regime)
- Heavily Doping  $\Rightarrow S = a \times T$   
 $\Downarrow$   
**Mott Relation**



Mott relation of thermopower

$$S = \left( \frac{\pi^2 k^2 T}{3 q} \right) \frac{1}{\sigma_E} \frac{d\sigma_E}{dE} \Big|_{E=E_F}$$

where  $\sigma_E$  is the *transport function*

$$\sigma_E = \frac{q^2}{3} \tau(E) v^2(E) N(E)$$

# Can We Improve The Thermopower?

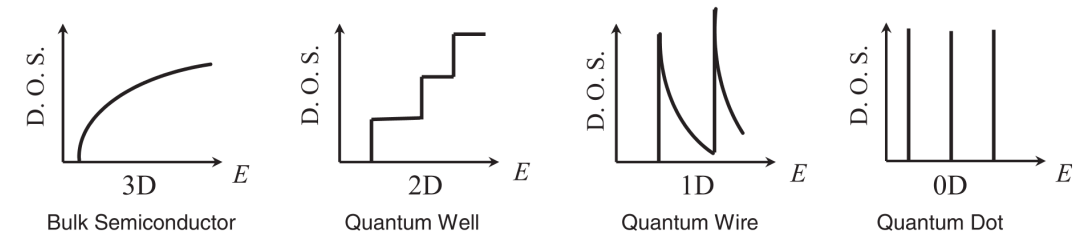
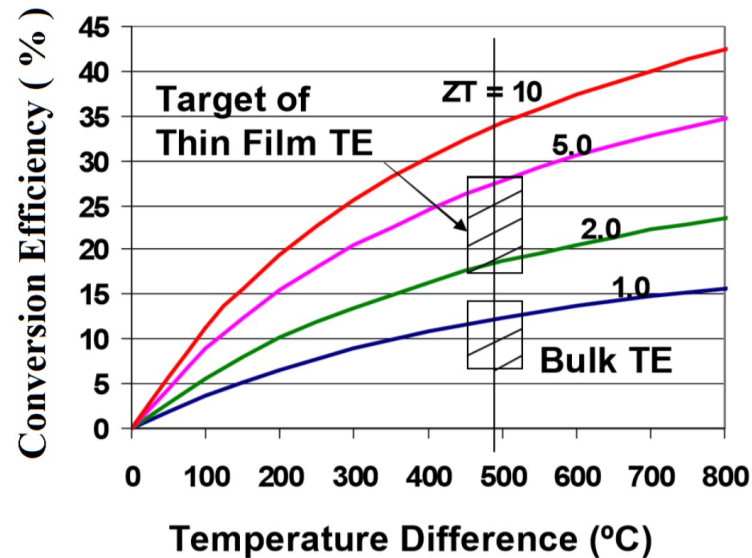
From the Mott formula:

$$S = \left( \frac{\pi^2 k^2 T}{3q} \right) \frac{1}{\sigma_E} \frac{d\sigma_E}{dE} \Big|_{E=E_F} \text{ and } \sigma_E = \frac{q^2}{3} \tau(E) v^2(E) N(E)$$
$$\frac{d\sigma_E}{dE} \approx \frac{dN(E)}{dE}$$

$\Rightarrow$  if we can *make the DOS increase substantially near the Fermi level*, then S may increase notably

# Can We Improve The Thermopower?

## 1) Design of *low-dimensional* materials



**Figure 1.** Electronic density of states for a) a bulk 3D crystalline semiconductor, b) a 2D quantum well, c) a 1D nanowire or nanotube, and d) a 0D quantum dot. Materials systems with low dimensionality also exhibit physical phenomena, other than a high density of electronic states (DOS), that may be useful for enhancing thermoelectric performance (see text).

From: Dresselhaus et al., Adv. Mater. 2007, 19, 1-12

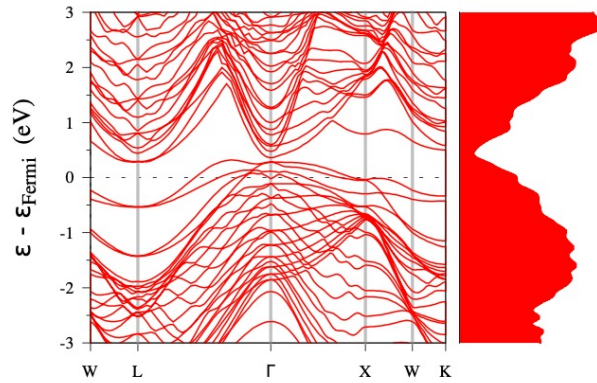
2) *Merge bands* near the Fermi level by applying *strains*

3) Introduce foreign atoms to *create resonant levels*

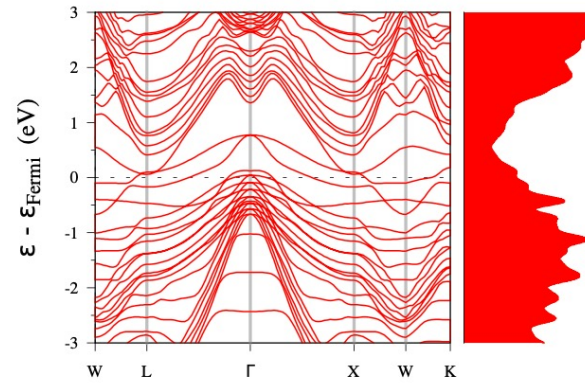
Examples with  $\text{Mg}_2\text{Si}$  and HMS

# Band Engineering in $\text{Mg}_2\text{Si}$ : Band Structures

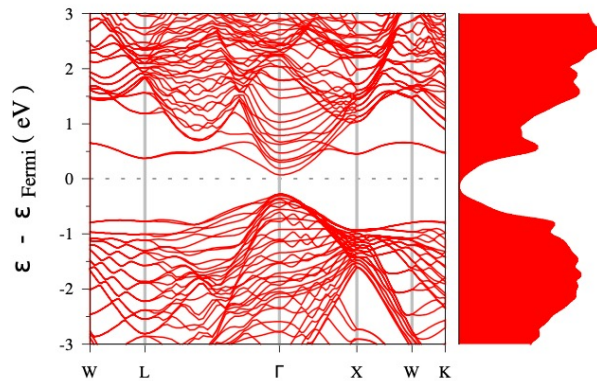
(001)



(111)



(110)



110 thin Film is a  
Semiconductor

$$E_{gap}^{16AL} = 274 \text{ meV}$$

## Comments

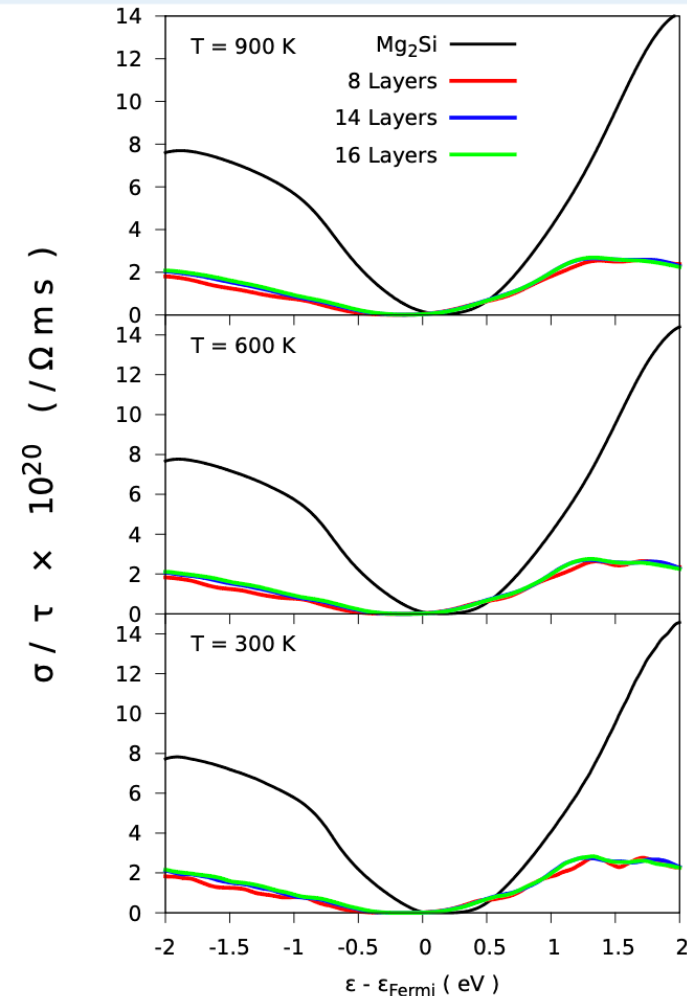
- Only the (110)-oriented thin film is SC
- Flat bands localized in the conduction bands of the (110) film
- Energy gap slightly larger than that of the bulk
- Energy gap evolves with the number of layers

# Band Engineering in $\text{Mg}_2\text{Si}$ : Electrical Conductivity

- $\sigma$  Curves of the Slabs Are Almost Superimposed
- Temperature Marginally Affects  $\sigma$
- $0 \leq \Delta E \leq 0.5 \text{ eV} \Rightarrow$  Thin Film Exhibits the Highest  $\sigma$



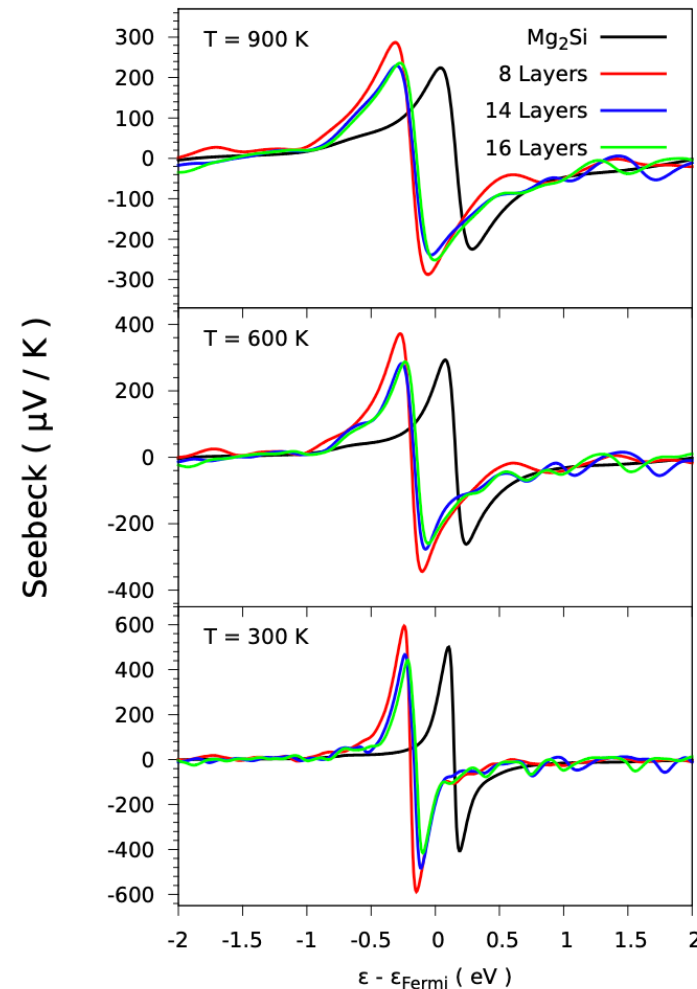
**(110)  $\text{Mg}_2\text{Si}$  Thin Film Have Better Conduction then Bulk**





# Band Engineering in $\text{Mg}_2\text{Si}$ : Seebeck Coefficient

- **Shift (Between Films and Bulk) Results from the Fermi Level Positions**
- **The Thinner the Films, the Larger the S**
- **S Increases Strongly when the Fermi Level Falls into Gap**
- **Temperature Affects Sensibly S**

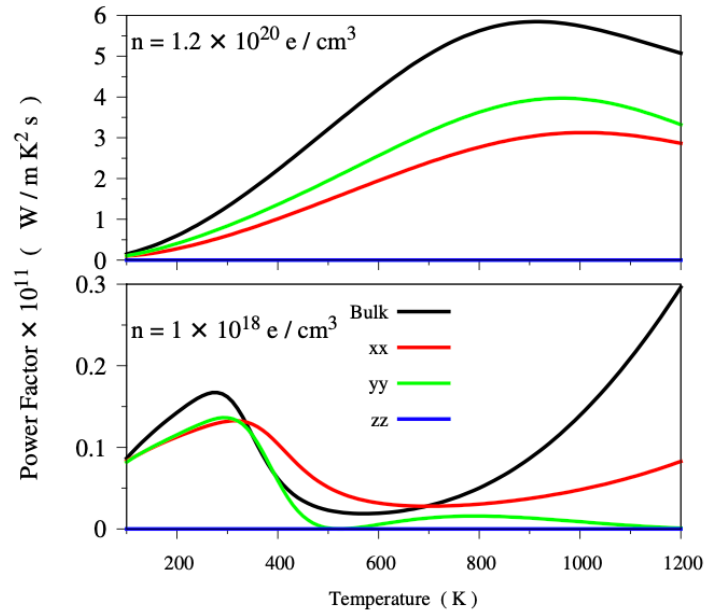


# Band Engineering in $\text{Mg}_2\text{Si}$ : Power Factor

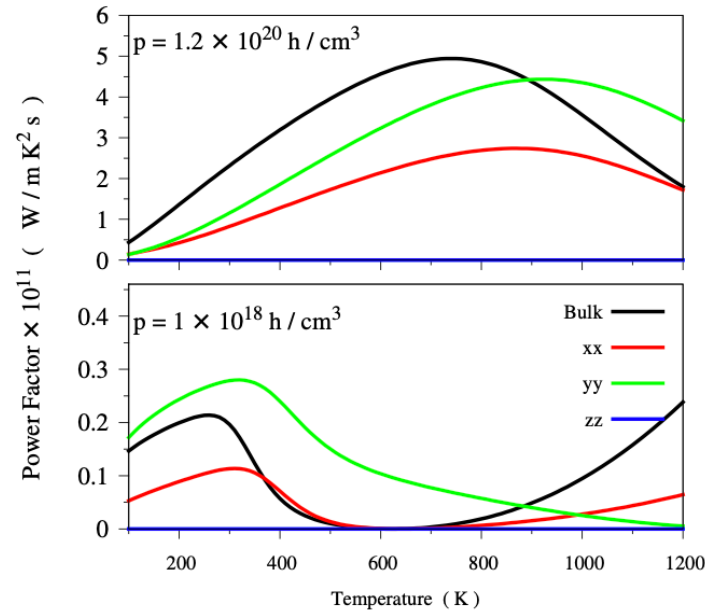
- **Anisotropy in Structure  $\Rightarrow$  Anisotropy in TE Properties.**
- **$\text{PF}_{xx}$  Highest for Low n-Doped Thin Film for  $350 \leq T \leq 650$  K**
- **$\text{PF}_{yy}$  Highest for Low p-Doped Thin Film for  $T \leq 850$  K**

## Conclusions

- Both  $S$  and  $\sigma$  can be improved
- Idea of decoupling  $S$  and  $\sigma$  through band engineering
- $S$  is highest for the thinnest film
- Best performance at low doping and moderate  $T$



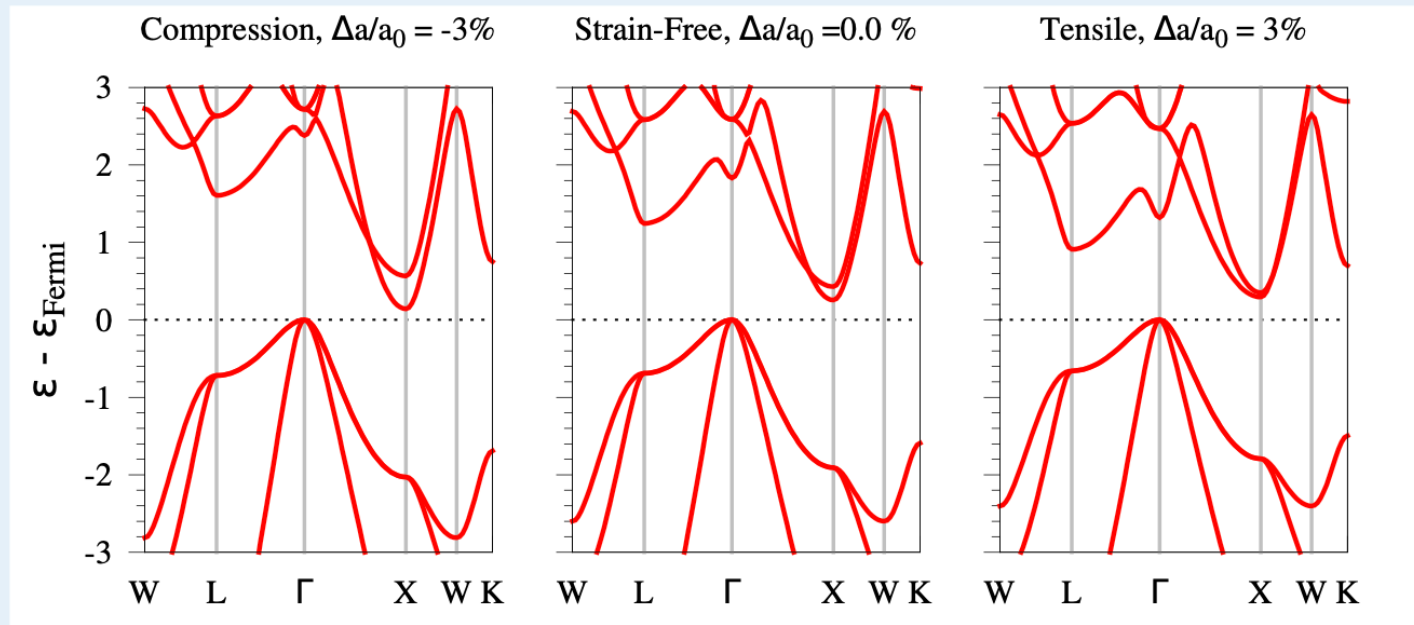
n-Type



p-Type

# Band Engineering in $\text{Mg}_2\text{Si}$ : bands Convergence

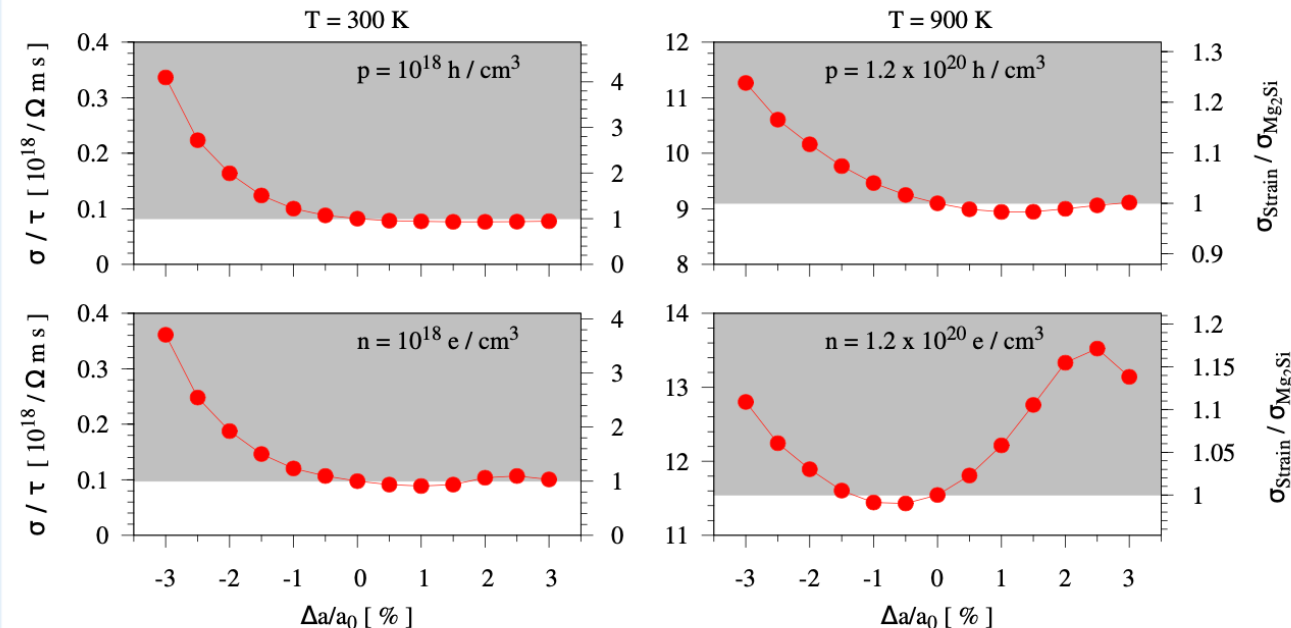
✠ Isotropic Strain  $\Rightarrow$  Changing  $a$  ( $\Delta a/a_0$  going from -3% to +3%)



- NO Removing of Bands Degeneracy.
- Compressive Strain Tends to Close up the Band Gap.
- Tensile Strain Tends to Enlarge the Band Gap.

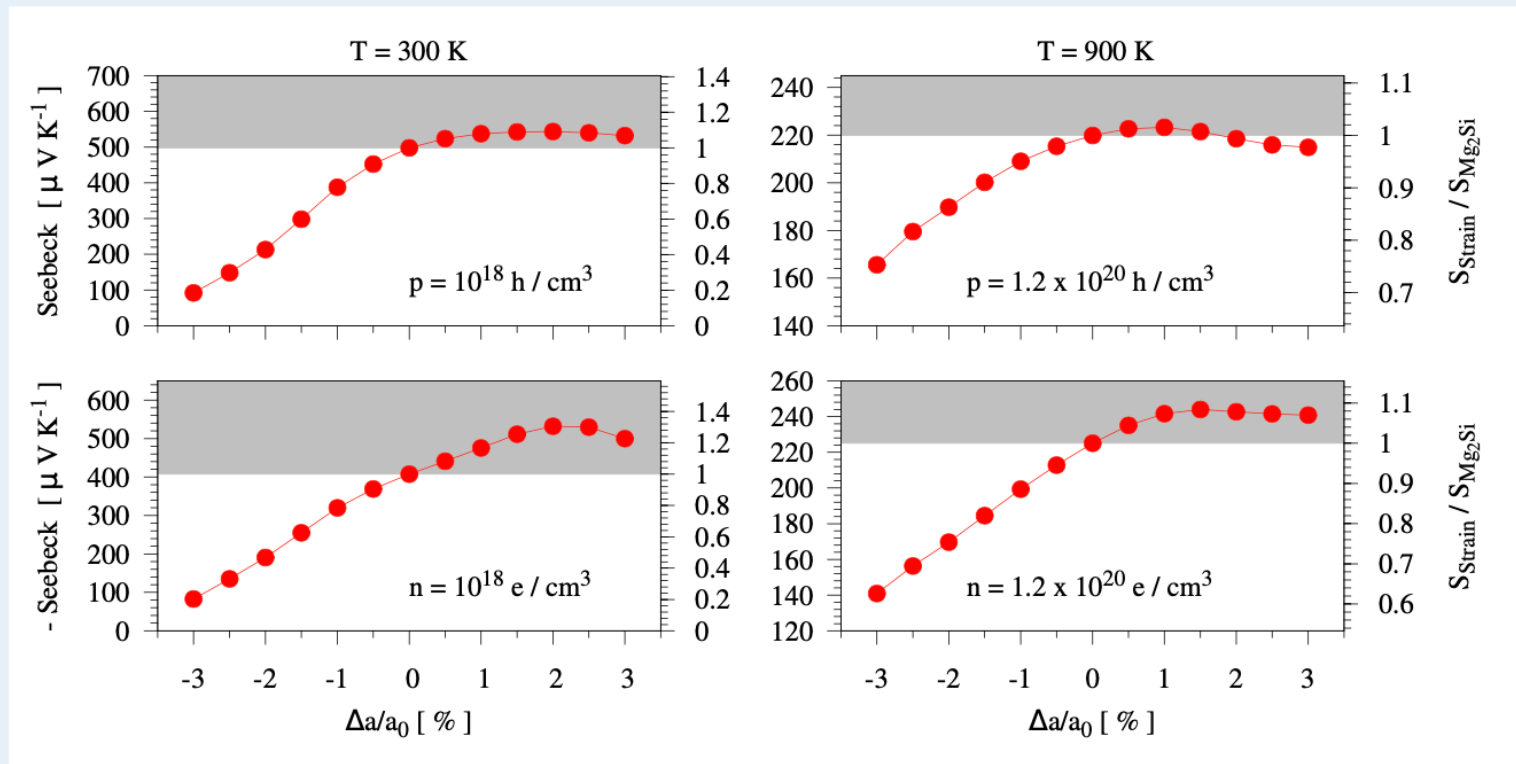
# Band Engineering in $\text{Mg}_2\text{Si}$ : Electrical Conductivity

- The Higher the Compression, the Higher the  $\sigma$
- Tensile Strain Affects Marginally  $\sigma$ , for p-type and Low n-Type Doping Level
- At High Electron Doping Level, Tensile Strain Enhances  $\sigma$  ( at 2.5 %)
- All of these Features Depend on the Magnitude of the Band Gap Energy



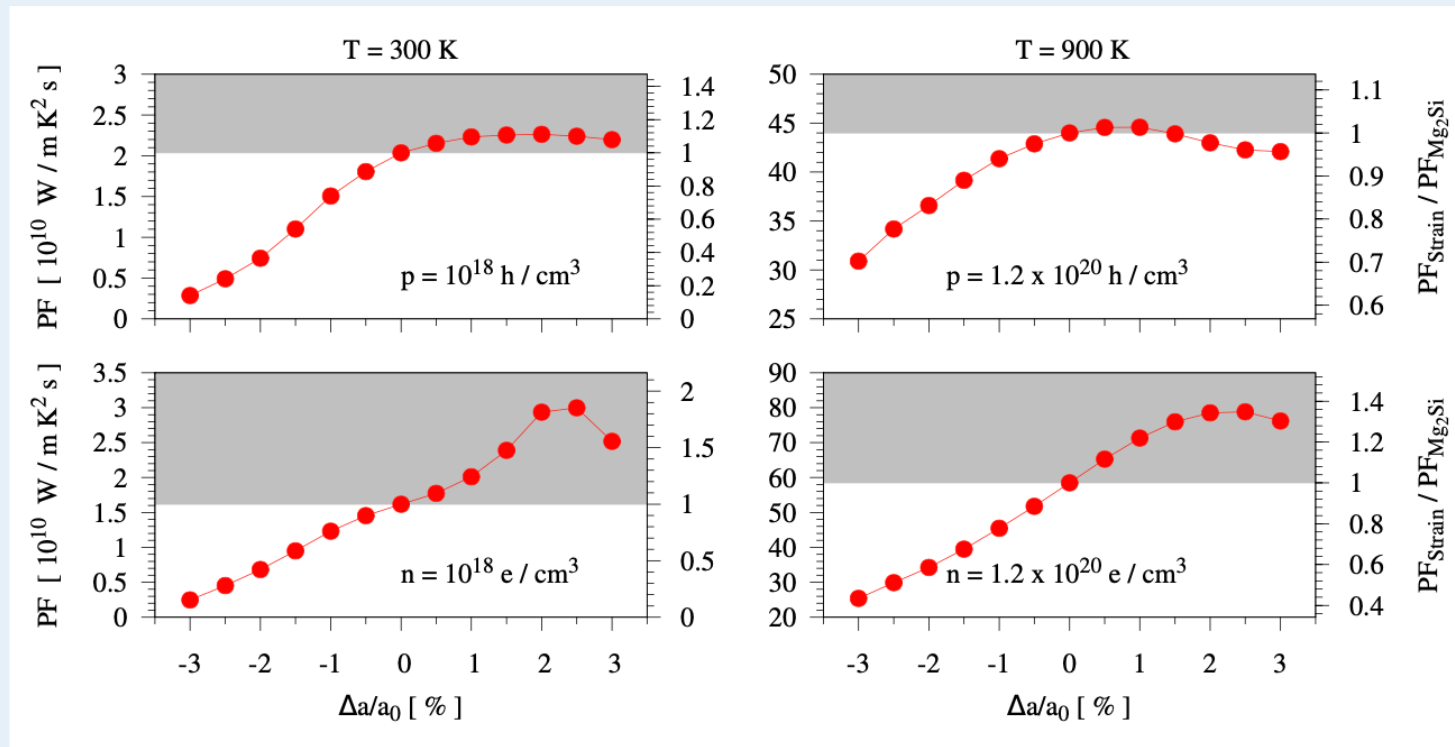
# Band Engineering in $\text{Mg}_2\text{Si}$ : Seebeck Coefficient

- **S Decreases Monotonically Under Compressive Strain**
- **Tensile Strain Enhances S, Specially at Low Doping Level  $10^{18} \text{ cm}^{-3}$**



# Band Engineering in $\text{Mg}_2\text{Si}$ : Power factor

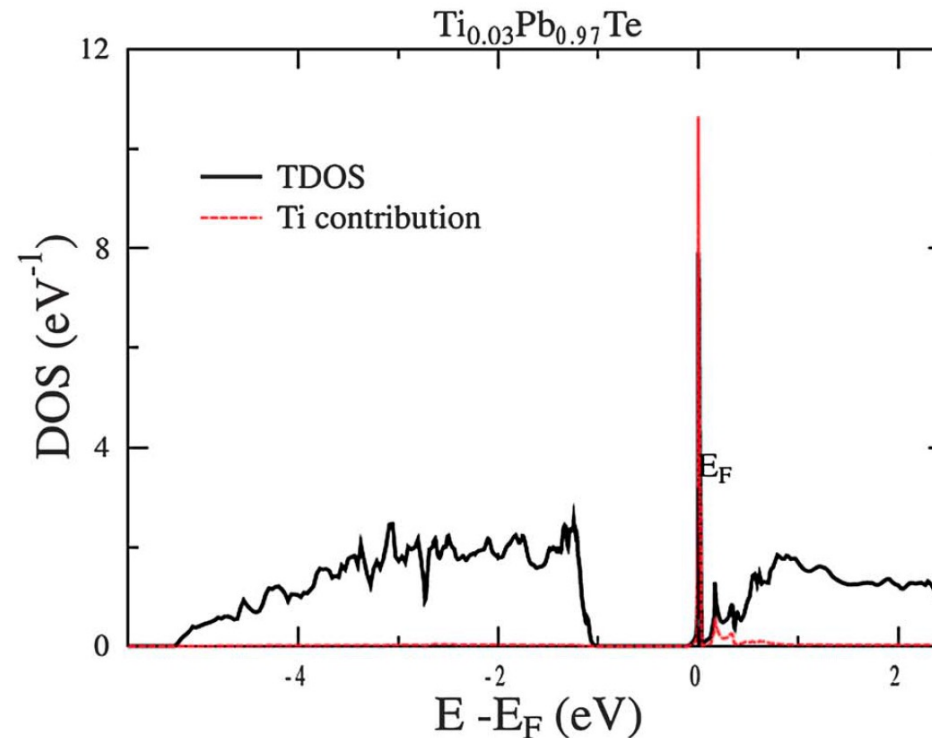
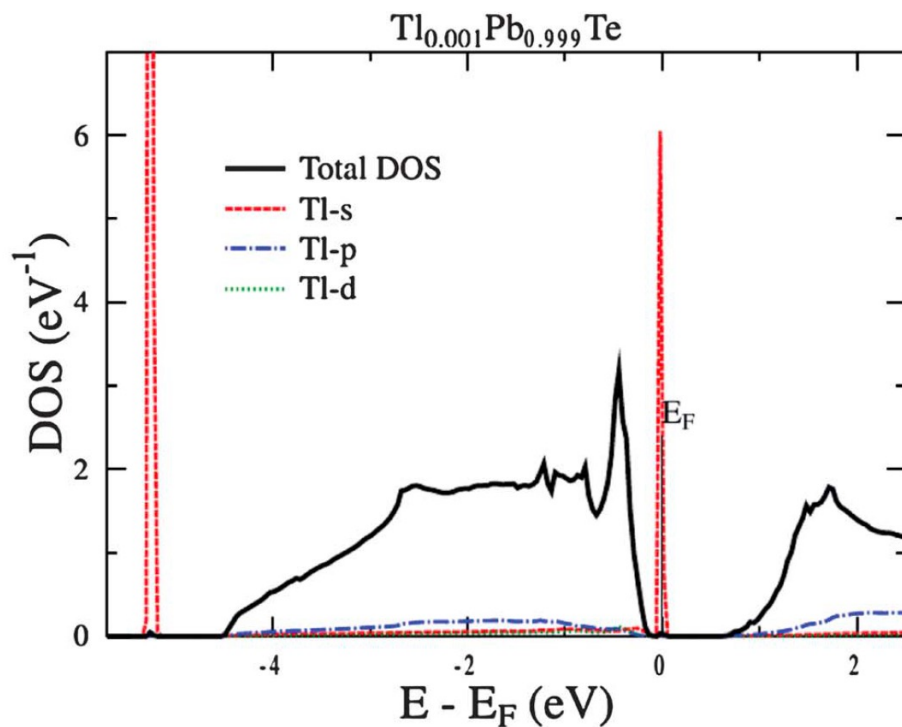
- Since  $S \propto 1/\sigma \Rightarrow \text{PF} = \text{Compromise Between } S \text{ and } \sigma$
- Under Tensile Strain  $\left\{ \begin{array}{l} \text{Low e-doping : } \sigma \simeq \text{cte and Higher } S \Rightarrow \text{Higher PF.} \\ \text{High e-doping : Higher } \sigma \text{ and Higher } S \Rightarrow \text{Higher PF.} \end{array} \right.$



# Band Engineering: Resonant Levels

What is a *resonant level*?

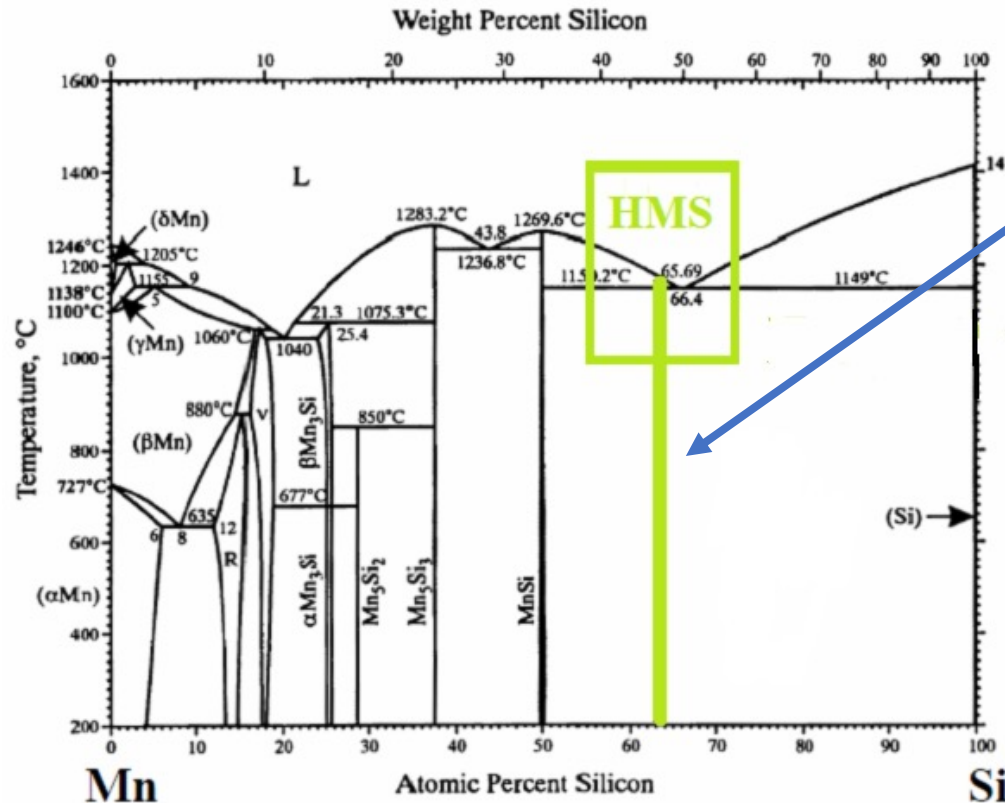
Energy states located at or near  $E_F$  in the DOS due to impurities



From:  
J. P. Heremans et al.,  
Energy Environ. Sci., 2012, 5, 5510

# Higher-Manganese Silicides

## Higher Manganese Silicides



Homogeneity range

63.2 at.%Si – 63.6 at.%Si

$\text{MnSi}_{1.72} - \text{MnSi}_{1.75}$



# Higher-Manganese Silicides

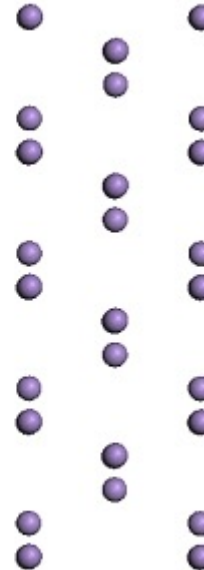
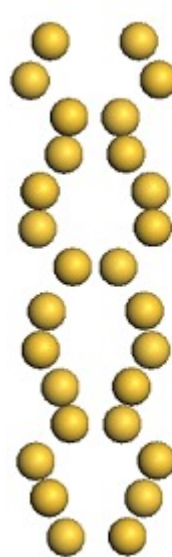
*Nowotny Chimney-ladder phases ( $a \approx 5.2 \text{ \AA}$ )*

	$\text{Mn}_4\text{Si}_7$	$\text{Mn}_{11}\text{Si}_{19}$	$\text{Mn}_{15}\text{Si}_{26}$	$\text{Mn}_{27}\text{Si}_{47}$
$X (\text{MnSi}_x)$	1.75	1.727	1.733	1.7407
$C (\text{\AA})$	17.45	48.13	65.55	117.9

Manganese



Silicon

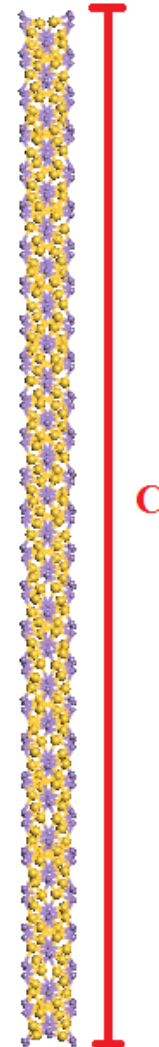


$\text{Mn}_4\text{Si}_7$

$\text{Mn}_{11}\text{Si}_{19}$

$\text{Mn}_{15}\text{Si}_{26}$

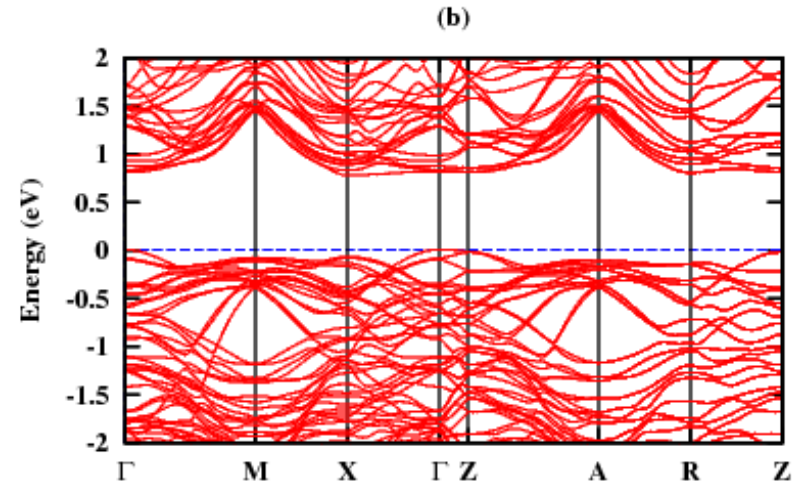
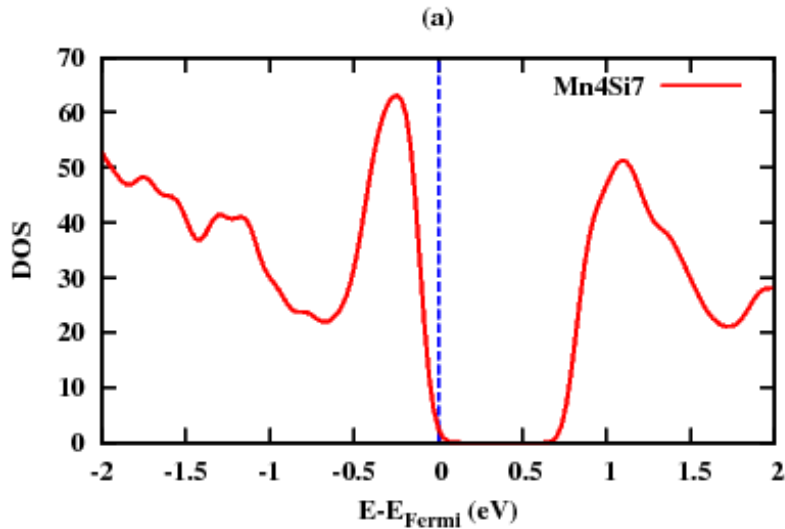
$\text{Mn}_{27}\text{Si}_{47}$



$c$

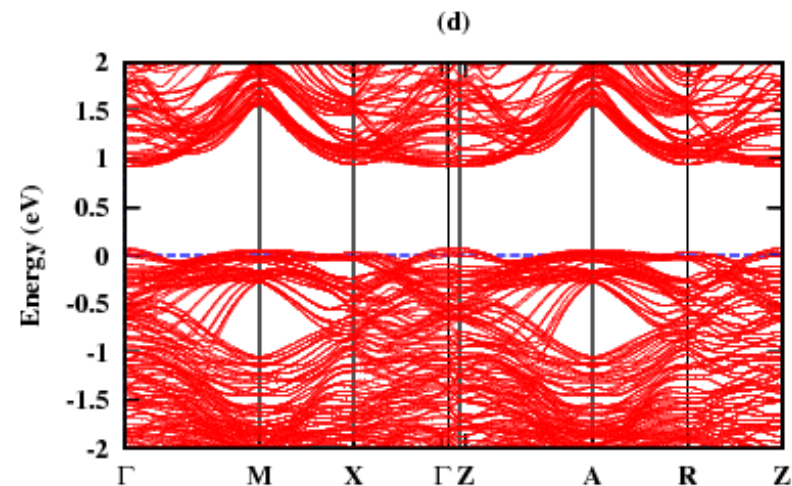
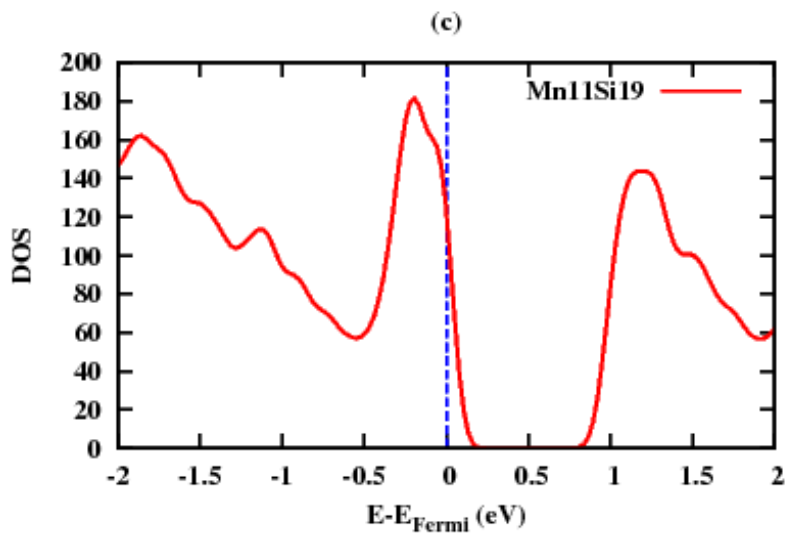


# Resonant levels in HMS



Mn<sub>4</sub>Si<sub>7</sub>

Semiconductor



Mn<sub>11</sub>Si<sub>19</sub>

Degenerate p-type SC

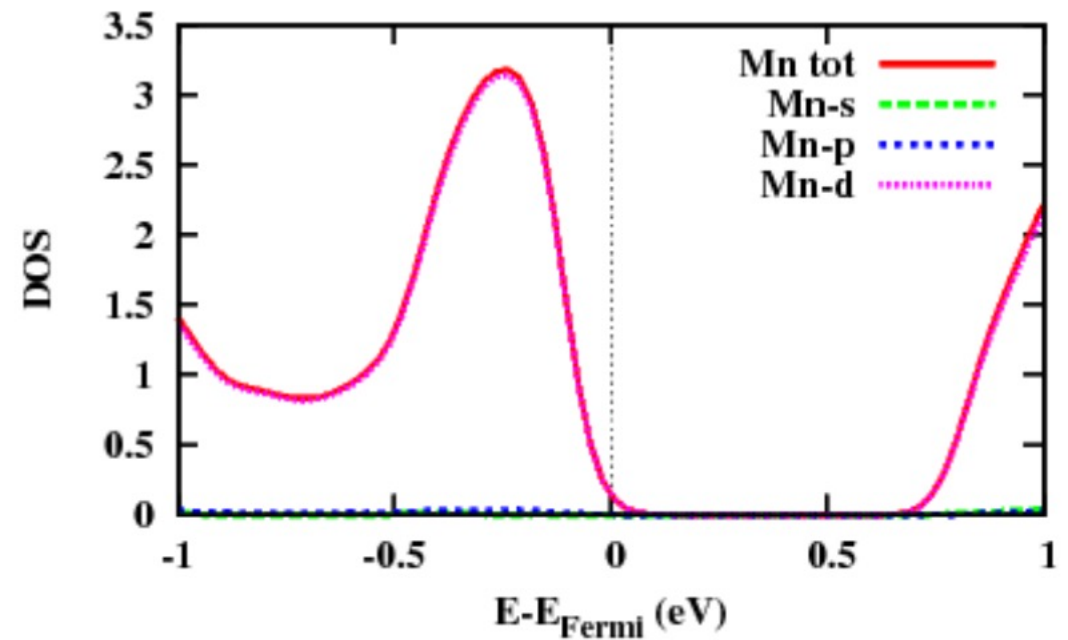
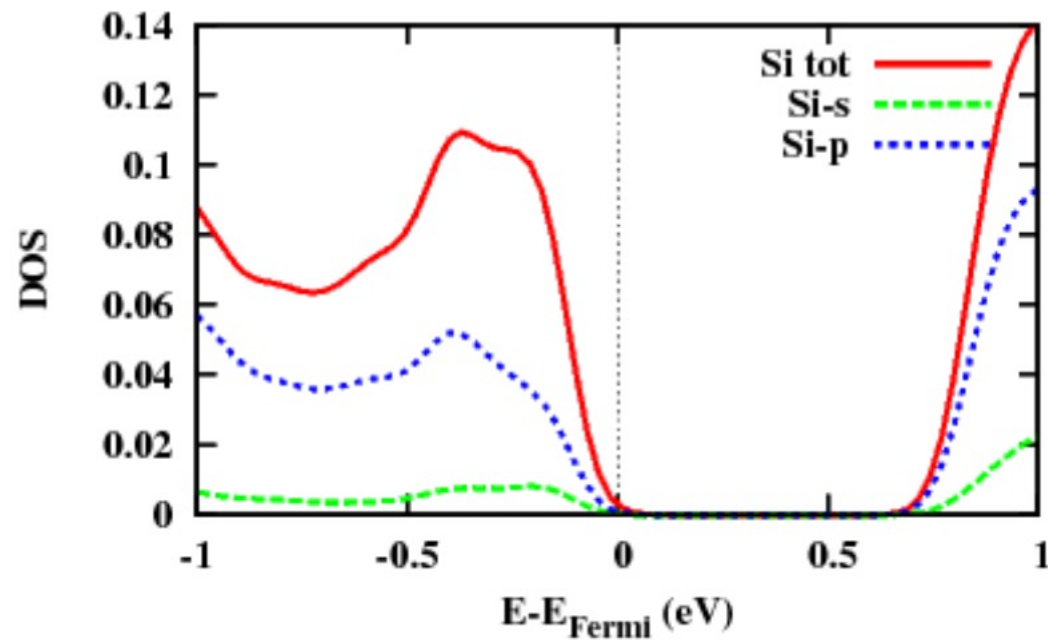
References:

Studies from A. Allam et al.

- J. Electron. Mater., 2014 (43) 761
- J. Alloy Compnd., 2014 (584) 279

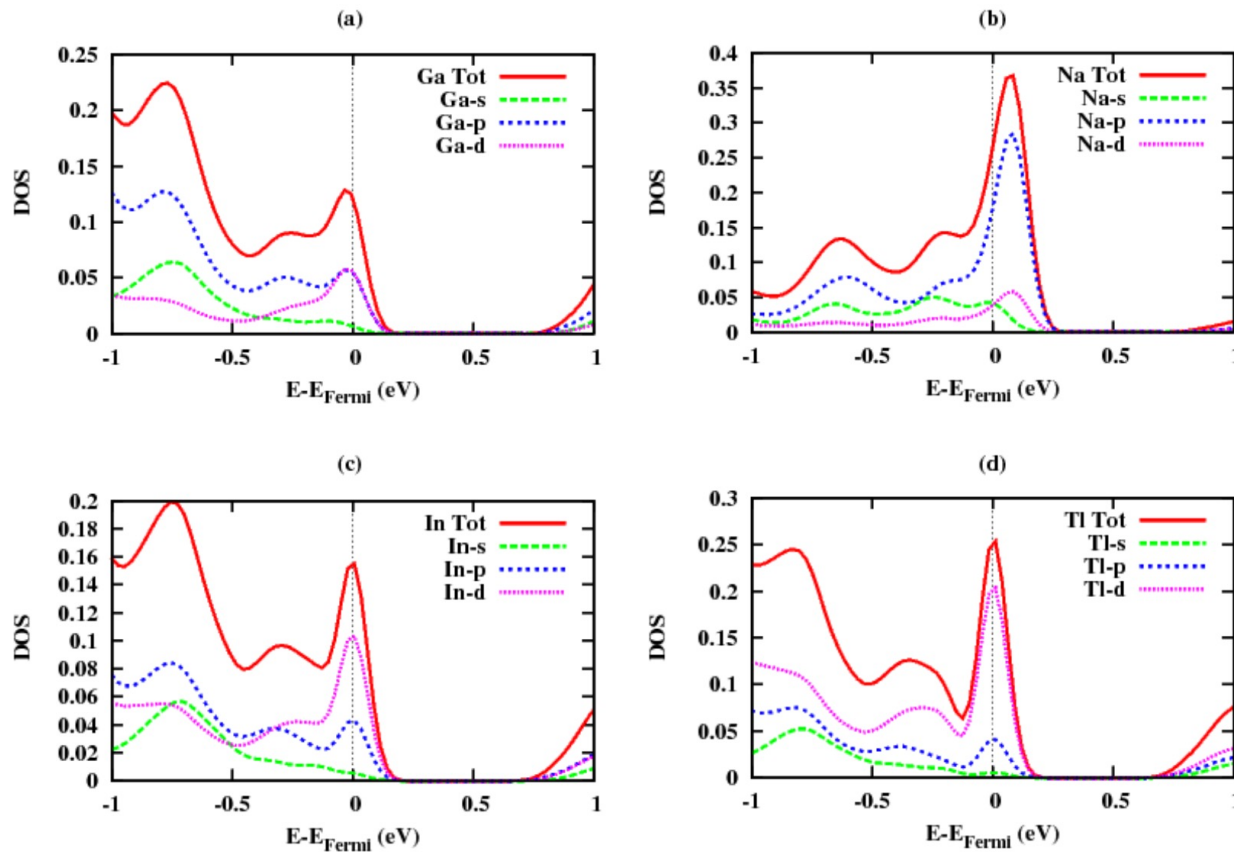
# Resonant levels in HMS

## Density of states of $\text{Mn}_4\text{Si}_7$ HMS



# Resonant levels in HMS

## Substitutions of 1.1at.% Ga, Na, In and Tl

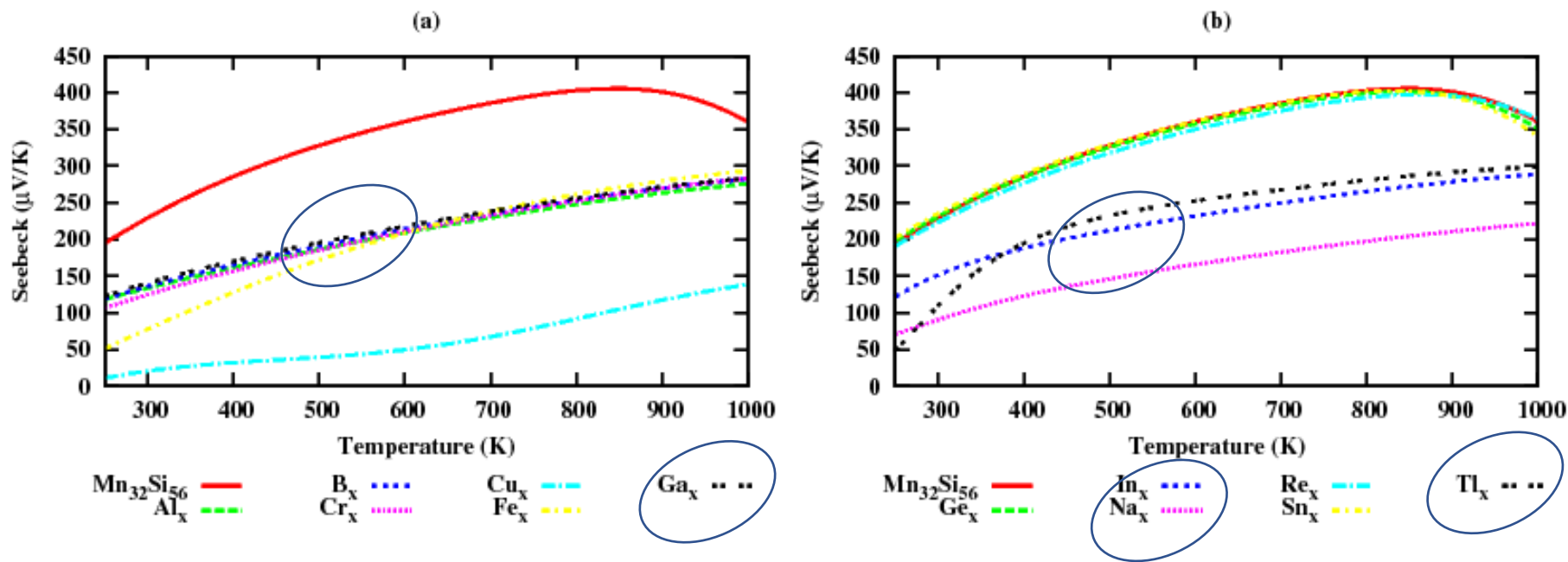


### Comments

- $\text{Mn}_4\text{Si}_7$  is p-doped
- Large peak appears near  $E_{\text{F}}$
- Peak caused by d-orbitals for In and Tl, mixed p/d-orbitals for Ga and p-orbitals for Na
- In the case of Na,  $E_{\text{F}}$  is deep in the valence band

# Resonant levels in HMS

- Seebeck coefficient of substituted  $\text{Mn}_4\text{Si}_7$



## Comments

- None of the candidates yield better Seebeck coefficient
- Ge/Sn: isoelectronic with Si
- Re: isoelectronic with Mn
- Cu: yields a metallic state

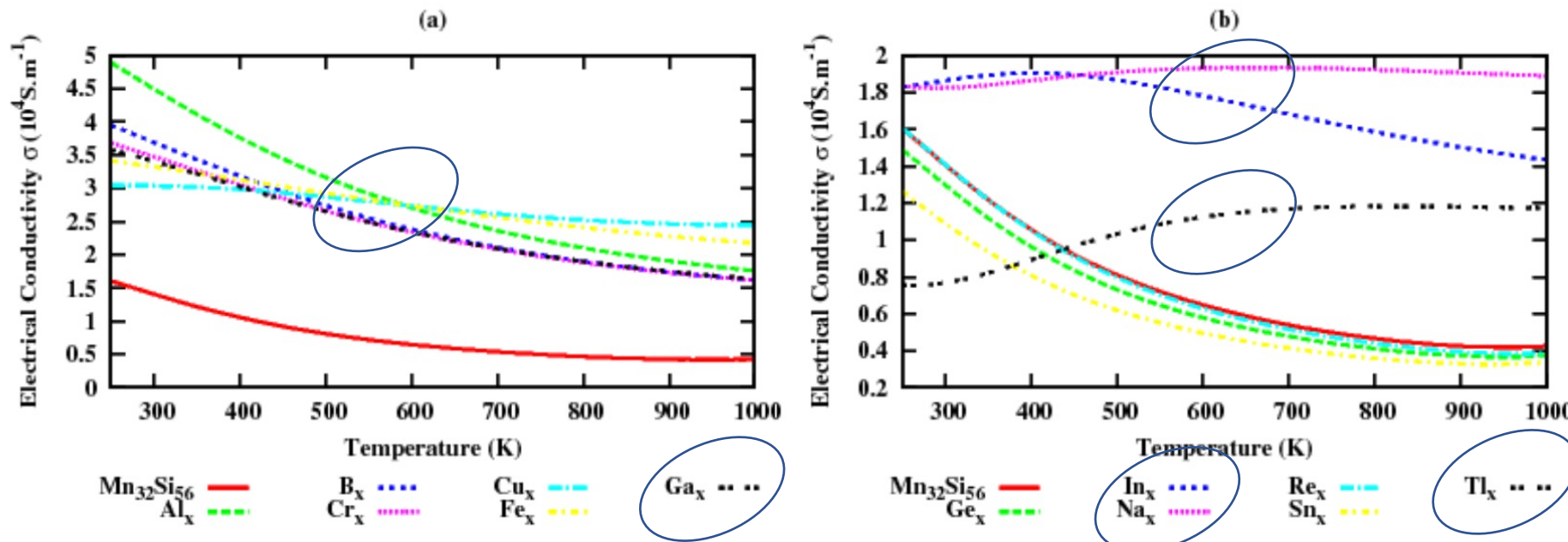
Why? {

- Peak width should be as small as possible
- The background DOS should be as small as possible
- In SC, for complex reasons, the states involved should be s or p

See J. P. Heremans et al., Energy Environ. Sci., 2012, 5, 5510

# Resonant levels in HMS

## Electrical conductivity of substituted $\text{Mn}_4\text{Si}_7$

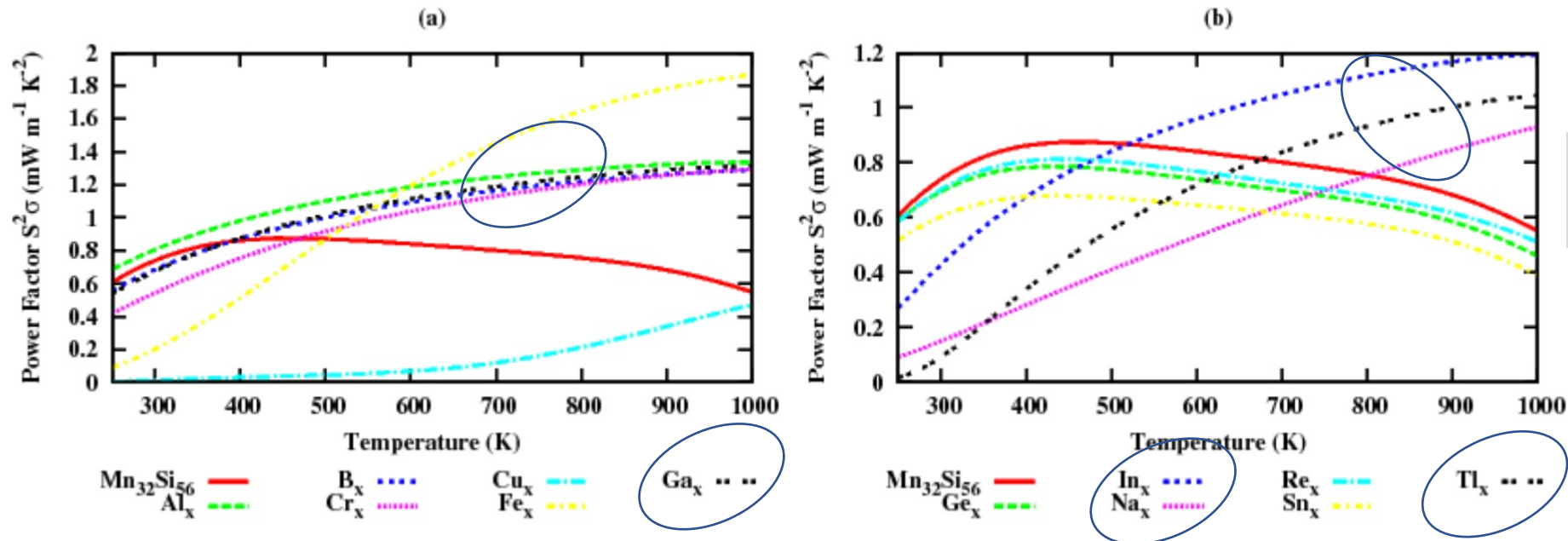


### Comments

- Conduction by holes where the Fermi level is, so high electrical conductivity of Ga-, Na-, In- and Tl-substituted  $\text{Mn}_4\text{Si}_7$
- Metallic compounds have inherently high electrical conductivity (Cu-, Fe-substituted  $\text{Mn}_4\text{Si}_7$ )



# Resonant levels in HMS



Comment

Power factor improved due to high electrical conductivity

## Conclusion

- Electronic thermoelectric properties of HMS could be improved by substitutions of impurities
- Not that clear that these impurities induce resonant levels in the compound
- Resonant levels are easy to identify in theoretical toy-models, but not that easy in "real" compounds

# Conclusions

- The Landauer approach is well suited to understand the physics of the phenomena (Seebeck coefficient, Peltier effect, electrical conductivity and electronic thermal conductivity)
- The Boltzmann approach is well suited to solve problems numerically, in particular when one wants to include magnetic field
- {DFT + BTE} approach is interesting as it yields adequate preliminary results but there are weaknesses such as the lack of the treatment of collisions and of the relaxation time. To go beyond, Monte Carlo is more appropriate
- Transport is quite demanding in terms of the quality of the DFT calculations, especially regarding the thin sampling of the Brillouin zone.